

Phonon entropy of alloying and ordering of Cu-Au

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Inelastic neutron scattering spectra were measured with a time-of-flight spectrometer on six disordered Cu-Au alloys at 300 K. The neutron-weighted phonon density of states was obtained from a conventional analysis of these spectra. Several methods were developed to account for this neutron weighting and obtain the phonon entropy of the disordered alloys. The phonon entropies of formation of disordered fcc Cu-Au alloys obtained in this way were generally mutually consistent, and were also consistent with predictions from a cluster approximation obtained from *ab-initio* calculations by Ozoliņš, Wolverton, and Zunger. We estimate a phonon entropy of disordering of $0.15 \pm 0.05 k_B/\text{atom}$ in Cu_3Au at 300 K. A resonance mode associated with the motions of the heavy Au atoms in the Cu-rich alloys was observed at 9 meV. An analysis of the resonance mode provided a check on the partial phonon entropy of Au atoms.

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I. INTRODUCTION

It is universally appreciated that most of the heat in solids goes into the vibrations of atoms, and the associated phonon entropy is large. For many years, however, it was unclear if different alloy phases had sufficient differences in phonon entropy to affect their relative thermodynamic stabilities. One of the earlier controversies involved Cu_3Au , where the temperature dependence of the electrical resistivity was used to argue that phonon entropy strongly affected the entropy of the chemical order-disorder transition,¹ whereas low-temperature measurements of elastic constants were used to argue that the difference in phonon entropy was negligible.² Neither measurement provided direct information on phonon entropy, however. Over the past decade the experimental and theoretical efforts have become more focused, and a number of cases were found where phonon entropy is significant in structural and chemical phase transitions in alloys.³⁻¹³ The origins of these differences in phonon entropy are now an active topic of research.

The Cu-Au system has long been a testbed for studies of alloy phases and order-disorder phase transitions. At three stoichiometric compositions, the stable phases at low temperature are ordered intermetallic compounds having the $L1_2$ structures for Cu_3Au and CuAu_3 , and $L1_0$ for CuAu . At elevated temperatures, above a critical temperature of 663 K (390 °C) for $L1_2$ for Cu_3Au for example, Cu and Au form a face-centered-cubic (fcc) single-phase solid solution at all compositions. The chemical order of the intermetallic compounds can be suppressed by rapidly quenching the alloys from high temperature. The effects of ordering on numerous properties have been reported, including elastic constants,² heat capacity¹⁴ and lattice dynamics¹⁵. A property of an ordered phase is typically compared to that of a disordered phase with the same composition. When doing so, it is often found that the properties of the disordered alloy are the more difficult to understand. Without the simplifications that follow from Bloch's theorem, and with local relaxations of atom positions, true *ab-initio* calculations are expensive and methodologically uncertain, but progress has been made.^{13,16-18} The energy difference between the ordered and

disordered phases can be estimated by modern electronic structure calculations, and the *configurational* entropy is well understood through cluster-variation calculations or Monte Carlo simulations. The phonon component of the transition entropy, $\Delta S_{\text{ph}}^{\text{d-o}} = S_{\text{ph}}^{\text{dis}} - S_{\text{ph}}^{\text{ord}}$, where $S_{\text{ph}}^{\text{dis}}$ and $S_{\text{ph}}^{\text{ord}}$ are the phonon entropies of the disordered and ordered phases, is a challenge for theoretical calculations. Ozoliņš, Wolverton, and Zunger used an inverse cluster-variation method with *ab initio* calculations of the lattice dynamics of ordered Cu-Au phases to obtain a small $\Delta S_{\text{ph}}^{\text{d-o}}$ value of $0.06 k_B/\text{atom}$ at 300K.⁸ In experimental work using low-temperature calorimetry, Nagel, Anthony, and Fultz obtained a value of $\Delta S_{\text{ph}}^{\text{d-o}}$ of $0.14 \pm 0.05 k_B/\text{atom}$ for Cu_3Au at 300 K.¹⁹ They also used a Born-von Kármán model with force constants from phonon dispersion measurements of the ordered and disordered phases¹⁵ to obtain a $\Delta S_{\text{ph}}^{\text{d-o}}$ of $0.23 k_B/\text{atom}$ at 300 K. This calculation of the phonon density of states (DOS) suffers seriously from the use of the virtual crystal approximation to interpret the disordered phase lattice dynamics, however.

In the present investigation, we used a time-of-flight inelastic neutron spectrometer to measure the differences in phonon scattering from disordered Cu_3Au , obtaining data matched for comparison with prior work on ordered Cu_3Au .²⁰ Unfortunately, neutrons are scattered up to 3.21 times more strongly by phonons whose dominant motions involve Cu atoms, compared to phonons involving motions of Au atoms. This difference is responsible for a "neutron-weight" distortion of the phonon DOS. Until recently, there has been little work to overcome this basic characteristic of phonon scattering. When the Born-von Kármán model can be utilized to simulate the lattice dynamics, as is the case for the ordered phases, it is possible to calculate the neutron weight correction as was done for $L1_2$ -ordered Cu_3Au (Ref. 20) and $B2$ -ordered NiTi.¹⁰ Sometimes it is possible to iterate the interatomic force constants to optimize the match between calculated and experimental inelastic scattering, and use the force constants so obtained for a neutron weight correction, as was done for $L1_2$ -ordered Pt_3Fe .²¹ Chemical disorder in solid solutions of Cu-Au removes translational peri-

odicity, precluding neutron-weight corrections with a Born-von Kármán model. The central focus of the present research is developing a reliable methodology for obtaining the phonon *entropy* from the neutron-weighted phonon DOS, and not the phonon DOS itself. Several neutron-weight corrections are developed and tested, including a comparison with effective cluster interactions from an *ab initio* theoretical study. We report phonon entropies of formation of disordered Cu-Au alloys, and the consistency of these results with each other and with theoretical results⁸ indicates their reliability. We also report a difference in phonon entropy of disordered and ordered Cu₃Au of $0.15 \pm 0.05 k_B/\text{atom}$ at 300 K.

II. EXPERIMENTAL METHODS AND RESULTS

Pieces of Au and Cu, both of 99.99% purity, were arc-melted to provide ingots with composition Cu_xAu_{1-x}, where the values of x were 96, 82, 75, 68, 50, and 25 %. The total mass lost during arc-melting was negligible, and there was no evidence of a surface oxide. Each ingot was cold-rolled to a thickness of approximately 0.76 mm. The cold-rolled strips were cut into smaller pieces, annealed for 45 min at 1123 K (850 °) in evacuated quartz ampoules, then quenched into iced brine to preserve chemical disorder. X-ray diffractometry with Co $K\alpha$ radiation and an Inel CPS-120 position sensitive detector showed no evidence of chemical long-range order in the quenched specimens.²⁰ The presence of short-range order (SRO) is harder to rule out, but the quenched samples showed no modulations in the background around the positions of the superlattice diffractions. A slow quench from above the critical temperature can produce samples with partial SRO however,²² typically when the quenched samples have equiaxed shapes of approximately 1 cm dimension.²³ Cooling rates of hundreds of degrees K per second largely suppress changes in SRO in Cu₃Au.²⁴ Such rates may be achieved by our quenching method, which did succeed in producing significant differences between the ordered and disordered samples of Cu₃Au, as reported below.

Inelastic neutron scattering spectra were measured with the time-of-flight (TOF) chopper spectrometer LRMECS, at the IPNS spallation neutron source at Argonne National Laboratory. All samples were assembled as a mosaic plate of 7×9 cm, and encased in thin-walled Al pans for mounting in the displax refrigerator of the LRMECS spectrometer. The plates and powder samples were mounted at a 45° angle normal to the incident beam to minimize self shielding. Spectra with an incident energy E_i of 35 meV were measured at 300 K for all samples and an empty container. A run with a “black absorber” cadmium sheet in the sample position was measured at room temperature to improve the background correction. Detector response was calibrated with a run performed on a vanadium sample at room temperature, using broadband radiation without a Fermi chopper. The continuous detector coverage on LRMECS, ranging from 3° to 117°, allowed a simultaneous probe of the spectral response over a large range of momentum and energy transfer

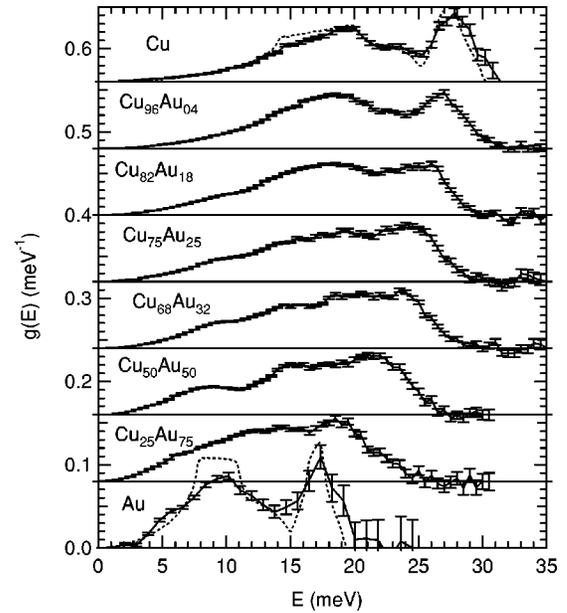


FIG. 1. Neutron-weighted phonon DOS of disordered Cu_{1-x}Au_x alloys. Thin dashed curves for elemental Cu and Au were calculated using force constants of Ref. 34 and 18.

($\vec{Q}, \hbar\omega$), with $|\vec{Q}(\omega)|$ varying between 0.2 and 7 Å⁻¹ at zero energy transfer.

All TOF spectra were first normalized and corrected for detector efficiency and time-independent background following standard procedures. In this step, the detectors were combined into 12 groups spanning 10° each with average angles from 5° to 115°. Scattering from the container and spectrometer hardware was determined using the runs on the empty container and Cd sample, taking into account the sample absorption. The data below 3–4 meV are dominated by the large elastic peak, which was stripped from the data. The inelastic scattering at these low energies is featureless and linear in the hydrodynamic limit, and is estimated easily. The neutron-weighted phonon density of states was determined from the corrected spectra with the procedure described previously.²⁰ This procedure is an iterative one, where a DOS curve is assumed, from which the single phonon and multiphonon scatterings are then calculated. Subtracting the multiphonon scattering from the background-corrected data leaves the single-phonon scattering, from which a new phonon DOS is obtained by a thermal correction. Typically the data require only two iterations before self-consistency is found for the DOS curve (or for alloys the neutron-weighted DOS curve).

The final neutron-weighted phonon DOS curves of Cu_{1-x}Au_x are shown in Fig. 1. These curves change continuously with decreasing Cu concentration. Two changes are most evident. The DOS softens monotonically with increasing Au concentration. This change can be seen most clearly in the high energy cutoff of the DOS, which is 31 meV for Cu and 20 meV for Au. The effect is not quite linear in Au concentration, as can be observed in Fig. 1—the DOS of Cu₅₀Au₅₀ terminates at 27 meV, not at 25.5 meV. Second, a peak at 9 meV grows rapidly with increasing Au concentra-

tion. This peak is almost certainly a resonance mode due to Au vibrations.

III. DISCUSSION

A. Neutron weighting

The experimental curves of Fig. 1 could be distorted by neutron weighting from the shapes of the true phonon DOS curves, perhaps significantly. The ratios $\sigma_{\text{Cu}}/m_{\text{Cu}}$ and $\sigma_{\text{Au}}/m_{\text{Au}}$, which are the most important factors controlling the relative efficiencies of phonon scattering by Cu and Au atoms, differ by a factor of 3.21. This implies that some parts of the neutron-weighted DOS, if associated with motions of Au atoms, could be underweighted by a factor as large as 3.21 compared to the true phonon DOS. Furthermore, the neutron-weight correction is different for each of the six alloys. (There is, of course, no neutron weight correction for the pure elements Cu and Au.)

In the quasi-harmonic approximation, the phonon DOS at temperature T and volume V , $g_{T,V}(E)$, provides the phonon entropy, $S_{\text{ph}}(T)$,

$$S_{\text{ph}}(T) = -3k_{\text{B}} \int_0^{\infty} g_{T,V}(E) [(n_E + 1) \ln(n_E + 1) - n_E \ln(n_E)] dE, \quad (1)$$

where n_E is the phonon occupancy factor. Unfortunately, we do not measure $g_{T,V}(E)$ directly, but spectra as in Fig. 1 are neutron-weighted DOS curves, $g_{T,V}^{\text{nw}}(E)$. This $g_{T,V}^{\text{nw}}(E)$ is composed of neutron-weighted contributions from the partial DOS curves of elements, $g^d(E)$, where “ d ” denotes Cu or Au,

$$g_{T,V}^{\text{nw}}(E) \propto \exp(2W) \sum_d g^d(E) \exp(-2W_d) \frac{\sigma_d}{m_d}, \quad (2)$$

and $\exp(-2W_d)$ is the Debye-Waller factor of atom d . The factor $\exp(2W)$ is the average Debye-Waller correction calculated from the self-consistent neutron-weighted DOS. In what follows we assume $\exp[-2(W - W_d)] = 1$, an assumption that is most reliable at low temperatures. In this case we can write simply

$$g_{T,V}^{\text{nw}}(E) = C \sum_d g^d(E) \frac{\sigma_d}{m_d} x_d, \quad (3)$$

where the normalization, C , depends on concentrations of the elements and their scattering strengths,

$$C = \left(\sum_d \frac{\sigma_d}{m_d} x_d \right)^{-1}. \quad (4)$$

If we evaluate Eq. 1 using the neutron-weighted DOS, $g_{T,V}^{\text{nw}}(E)$, instead of the true phonon DOS, $g_{T,V}(E)$, we obtain a “neutron-weighted phonon entropy,” $S_{\text{ph}}^{\text{nw}}(T)$, where

$$S_{\text{ph}}^{\text{nw}}(T) \equiv -3k_{\text{B}} \int_0^{\infty} g_{T,V}^{\text{nw}}(E) [(n_E + 1) \ln(n_E + 1) - n_E \ln(n_E)] dE. \quad (5)$$

This $S_{\text{ph}}^{\text{nw}}(T)$ proves useful because of how it is related to the different $g^d(E)$. Substituting Eq. (3) into Eq. (5)

$$S_{\text{ph}}^{\text{nw}} = -3k_{\text{B}} C \sum_d \frac{\sigma_d}{m_d} x_d \int_0^{\infty} g^d(E) [(n_E + 1) \ln(n_E + 1) - n_E \ln(n_E)] dE, \quad (6)$$

$$S_{\text{ph}}^{\text{nw}} = C \sum_d \frac{\sigma_d}{m_d} x_d S_{\text{ph}}^d(T), \quad (7)$$

where $S_{\text{ph}}^d(T)$ is the phonon entropy of atom d .

The neutron-weighted entropy is thus a sum of the partial entropies, $S_{\text{ph}}^d(T)$, weighted by the phonon scattering strengths of the atoms, d . Equation (7) is analogous to the phonon entropy, which equals the sum of the phonon partial entropies of all species,

$$S_{\text{ph}}(T) = \sum_d S_{\text{ph}}^d(T), \quad (8)$$

where each partial phonon entropy, $S_{\text{ph}}^d(T)$, is related to a normalized phonon partial DOS, $g^d(E)$, in the usual way

$$S_{\text{ph}}^d(T) = -3k_{\text{B}} \int_0^{\infty} g^d(E) [(n_E + 1) \ln(n_E + 1) - n_E \ln(n_E)] dE. \quad (9)$$

For the case of the binary alloy $\text{Cu}_{1-x}\text{Au}_x$, Eqs. (8) and (7) are

$$S_{\text{ph}}(x) = (1-x) S_{\text{ph}}^{\text{Cu}}(x) + x S_{\text{ph}}^{\text{Au}}(x), \quad (10)$$

$$S_{\text{ph}}^{\text{nw}}(x) = C' (3.21(1-x) S_{\text{ph}}^{\text{Cu}}(x) + x S_{\text{ph}}^{\text{Au}}(x)), \quad (11)$$

where x is Au concentration, and $0 \leq x \leq 1$. Here the normalization constant C' is

$$C' = [3.21(1-x) + x]^{-1}. \quad (12)$$

In the present work we do not attempt to correct the $g_{T,V}^{\text{nw}}(E)$ because obtaining a phonon DOS would require an analysis of the lattice dynamics of the disordered alloy. Instead, we infer the true partial entropies $S_{\text{ph}}^{\text{Cu}}(x)$ and $S_{\text{ph}}^{\text{Au}}(x)$ of Eq. (10) from measured neutron-weighted entropies $S_{\text{ph}}^{\text{nw}}(x)$, and from these we obtain corrected phonon entropies, $S_{\text{ph}}(x)$.

B. Neutron deweighting in the point approximation and its modification

We seek true phonon entropies from neutron-weighted phonon entropies. The simplest chemical model of phonon entropy assumes

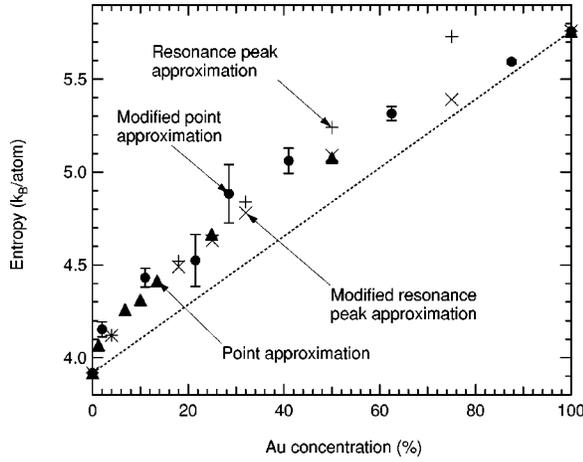


FIG. 2. Phonon entropies of disordered Cu-Au alloys at 300 K, corrected for neutron weighting by the methods indicated in labels. The straight dashed line is the straight line connecting the phonon entropies of elemental Au and Cu.

$$S_{\text{ph}}(x) = (1-x)S_{\text{ph}}^{\text{Cu}} + xS_{\text{ph}}^{\text{Au}}, \quad (13)$$

where neither $S_{\text{ph}}^{\text{Cu}}$ nor $S_{\text{ph}}^{\text{Au}}$ are functions of Au concentration, x . Equation (13) has a form similar to that of 10, and these two equations would be identical if $S_{\text{ph}}^{\text{Cu}}$ and $S_{\text{ph}}^{\text{Au}}$ were not functions of x . In this case the neutron-weighted phonon DOS at a particular composition would be identical to the true phonon DOS at a different composition (having more Cu). Compatibility of Eqs. (10) and (13) is then possible by defining a composition x^{nw} that weights more heavily the Cu atoms:

$$x^{\text{nw}} = \frac{x}{x + 3.21(1-x)}, \quad (14)$$

so that

$$S_{\text{ph}}^{\text{nw}}(x) = (1-x^{\text{nw}})S_{\text{ph}}^{\text{Cu}} + x^{\text{nw}} S_{\text{ph}}^{\text{Au}}. \quad (15)$$

Comparing Eqs. (15) and (13),

$$S_{\text{ph}}(x^{\text{nw}}) = S_{\text{ph}}^{\text{nw}}(x). \quad (16)$$

The phonon entropy, corrected in this ‘‘point approximation,’’ is shown in Fig. 2. The point approximation is strictly correct only if the partial entropies (and partial DOS) of Au and Cu are unchanged as the alloy varies in composition between x and x^{nw} . When there are large differences between x and x^{nw} , this approximation is likely to have large errors.

A ‘‘modified point approximation’’ improves upon the previous point approximation by measuring $S_{\text{ph}}^{\text{nw}}$ at two similar compositions, then solving two equations of the form of Eq. (15). The partial entropies, $S_{\text{ph}}^{\text{Cu}}$ and $S_{\text{ph}}^{\text{Au}}$ calculated this way are an average of similar partial entropies at the two alloy compositions. For example, using the neutron-weighted entropies of the $\text{Cu}_{68}\text{Au}_{32}$ and $\text{Cu}_{50}\text{Au}_{50}$ alloys, we calculate the average partial entropies $S_{\text{ph}}^{\text{Cu}} = 4.12k_{\text{B}}$ and $S_{\text{ph}}^{\text{Au}} = 6.41k_{\text{B}}$. From these values we calculate the phonon entropy of the alloy at composition $\text{Cu}_{59}\text{Au}_{41}$. The partial entropies of Au and Cu, calculated from pairs of adjacent com-

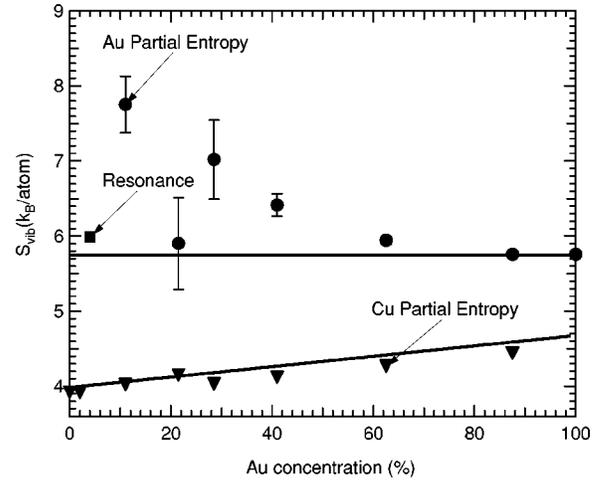


FIG. 3. Partial phonon entropies of Cu and Au atoms in Cu-Au alloys. Points were obtained in the modified point approximation, except point labeled ‘‘Resonance,’’ which was from the Au resonance peak approximation. [Lines were obtained with a three-parameter generalized partial entropy model. (Ref. 35)]

positions, are shown in Fig. 3. The closer are the two alloys in composition, the more accurate the assumption of equal partial entropies. Unfortunately, as the compositions become closer in value, it becomes more difficult to resolve accurately the difference in neutron-weighted entropies, given the statistical errors in the experimental spectra. The large error bars on the total entropy values at Cu concentrations of 78.5% and 71.5%, shown in Fig. 2, result from the small differences between the neutron-weighted entropies at alloy compositions $\text{Cu}_{82}\text{Au}_{18}$, $\text{Cu}_{75}\text{Au}_{25}$, and $\text{Cu}_{68}\text{Au}_{32}$. This is not a fundamental flaw with the approach, but it does require that the experimental data be of high quality.

C. Neutron deweighting with the local cluster approximation

The local cluster approximation (LCA) is used widely for calculating ground-state thermodynamic properties of chemically disordered materials.^{17,25} It can be used to parametrize the partial entropies of Cu and Au in disordered Cu-Au alloys. In the LCA the phonon entropy S_{vib} is

$$S_{\text{ph}} = S_0 + S_1 \sum_{\sigma_1} \phi(\sigma_1) + S_2 \sum_{\sigma_2} \phi(\sigma_2) + S_3 \sum_{\sigma_3} \phi(\sigma_3) + \dots, \quad (17)$$

where the S_i are the cluster coefficients associated with cluster i , σ_i is the i^{th} cluster type and $\phi(\sigma_i)$ is the spin-product over cluster σ_i . Cluster expansions have proved useful for analysis of previous experimental results.^{20,26} For $g_{\text{ph}}^{\text{nw}}(E)$ measured on fully disordered alloys, however, this parameterization cannot be used for reliable inversions to extract partial entropies. [For fully disordered alloys, the LCA basis functions become linearly-dependent beyond the point cluster. For example, probabilities for a first-nearest neighbor (1nn) pair are the same as for a 2nn pair.] On the other hand,

TABLE I. LCA cluster coefficients at 300 K from Ozoliņš, Wolverton, and Zunger. Units are in k_B/atom . The cluster expansion is for the phonon entropy of formation of disordered Cu-Au alloys. (Refs. 8 and 27)

	S_0	S_1	S_2	S_3	S_4
Cluster coefficients	0.234	0.0185	-0.108	-0.122	-0.0185

if the effective cluster interactions for phonon entropy are available, they can be tested against the experimental neutron-weighted DOS.

Ozoliņš, Wolverton, and Zunger⁸ recently performed *ab-initio* electronic-structure calculations of the phonon DOS and phonon entropy for a series of ordered Cu-Au alloys. Using results from six ordered Cu-Au alloys, they obtained a set of cluster-coefficients $\{S_i\}$ that best reproduced their results in Eq. (17). These clusters were the empty cluster S_0 , the point cluster S_1 , the 1nn pair S_2 , the 2nn pair S_3 and the 1nn triangle S_4 . The values for the cluster coefficients at 300K, given in Table I, were kindly provided by Ozoliņš.²⁷ Using the values given in Table I and Eq. (17), we can calculate the phonon entropy of formation of binary $\text{Cu}_{1-x}\text{Au}_x$ alloys of arbitrary Cu concentrations. For the case of a fully disordered binary system, Eq. (17) simplifies to

$$S_{\text{ph}}^{\text{form}} = S_0 + S_1(1-2x) + (S_2 + S_3)(1-2x)^2 + S_4(1-2x)^3. \quad (18)$$

Note that we include only five terms corresponding to the five clusters in our LCA sum. The terms corresponding to the 1nn pair and 2nn pairs, S_2 and S_3 , have the same functional dependence on x (because for fully disordered alloys the probabilities for 1nn and 2nn pairs are equal.) We evaluated Eq. (18) and show the results in Fig. 4. These LCA results are in generally good agreement with the results from our analyses of the experimental spectra.

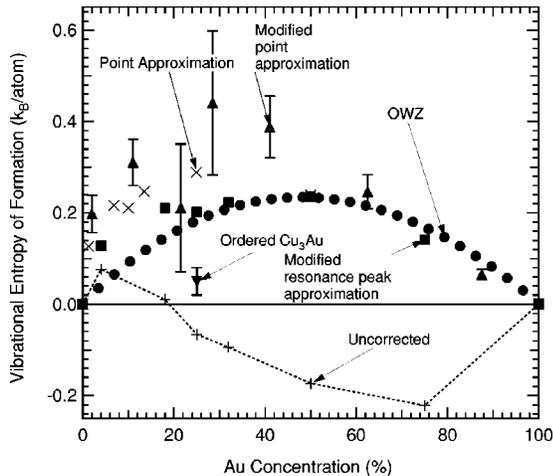


FIG. 4. Vibrational entropy of formation of disordered Cu-Au alloys at 300 K. Labels refer to neutron-weight corrections described in the text. The circles marked OWZ are the LCA results calculated with parameters used by Ozoliņš, Wolverton, and Zunger. (Refs. 8, and 27).

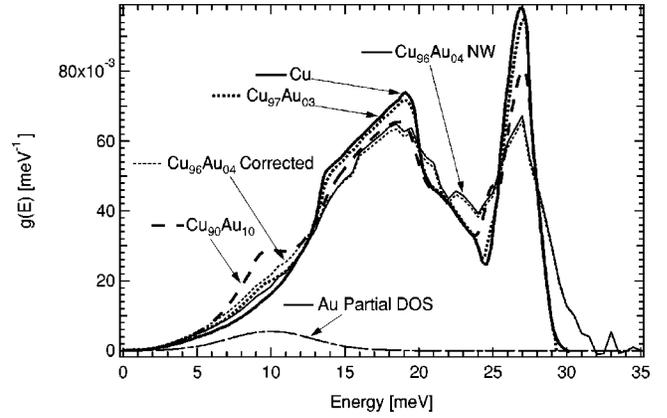


FIG. 5. Phonon DOS curves for Cu-Au alloys of dilute Au concentration. The experimental curves include the neutron-weighted DOS, Au partial DOS, and corrected DOS of $\text{Cu}_{96}\text{Au}_{04}$ alloy with the resonance peak approximation. The phonon DOS of 3% and 10% Cu-Au alloys were calculated by Elliott and Taylor (Ref. 33). The Cu curve was calculated using force constants of Ref. 18.

D. Au resonance mode and neutron deweighting

For Au concentrations between 0 and 50%, the peak at 9meV grows with increasing Au concentration. Similar phenomena have been reported in phonon dispersion measurements on Cu-Au (Ref. 28) and Cr-W (Refs. 29 and 30) dilute alloys. The low-energy peaks in dilute Cu-Au alloys are consistent with the resonance mode^{16,31} predicted by the mass-defect theory of lattice dynamics.³² A resonance mode comprises damped vibrational motions of heavy atoms, with contributions from the adjacent light atoms. With increasing concentration of the heavy atom, the partial DOS of the heavy atom broadens and develops more features. Elliott and Taylor³³ calculated the phonon DOS of 3% and 10% Cu-Au alloys using an analytical Green-function technique. Their results, presented in Fig. 5, show a low-energy resonance peak at 9 meV that grows proportionally with Au concentration. This peak was approximated adequately as a Gaussian function having a standard deviation of 2.5 meV. Using this peak for the Au partial DOS, $g^{\text{Au}}(E)$, we account for its neutron de-weighting and extract the Cu partial DOS, $g^{\text{Cu}}(E)$ from the measured $g_{T,V}^{\text{nw}}(E)$. [Although not shown in Fig. 5, the Cu partial DOS is nearly identical to the neutron-weighted DOS, $g_{T,V}^{\text{nw}}(E)$.] We then obtained the total DOS, $g(x,E)$, at each alloy composition. Results for the alloy composition $\text{Cu}_{96}\text{Au}_{04}$ are shown in Fig. 5. Using these total DOS curves, we obtained the phonon entropies shown in Figs. 2 and 4. At low Au concentrations, the entropies in the present “resonance-peak approximation” are in excellent agreement with the values from the point approximation and the modified point approximation. These results were largely insensitive to the width of the Gaussian that approximates the Au partial DOS.

The resonance-peak approximation for the Au partial DOS is a poor one for alloys with high Au concentrations. This is apparent by looking at the DOS of elemental Au in Fig. 1, which bears little resemblance to a Gaussian in energy centered at 9 meV. With a “modified resonance-peak approximation” we improved upon our scheme by assuming

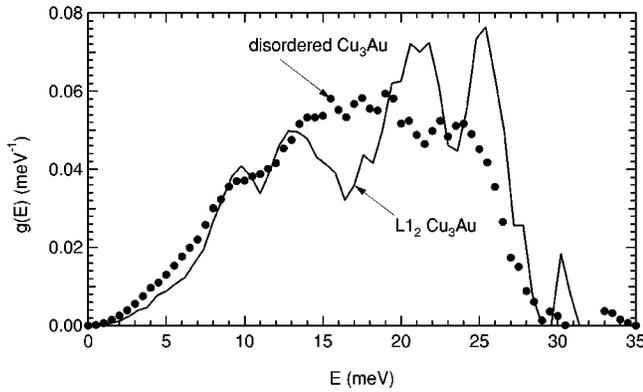


FIG. 6. Phonon DOS curves, corrected for neutron weighting of ordered and disordered Cu_3Au .

the Au partial DOS resembles the Gaussian resonance peak $G(E)$ at low Au concentrations, the elemental Au DOS at high Au concentrations, and a weighted sum of the two at intermediate Au concentrations,

$$g^{\text{Au}}(x, E) = (1-x)G(E) + xg_{\text{ph}}^{\text{Au}}, \quad (19)$$

where $g^{\text{Au}}(x, E)$ is the partial DOS of Au within an alloy of Au concentration x . The phonon entropies obtained from this modified resonance-peak approximation are shown in Figs. 2 and 4.

E. Vibrational entropy of ordering of $L1_2$ Cu_3Au

The phonon DOS of the ordered alloy can be obtained reliably because the neutron-weight correction can be calculated using the Born-von Kármán lattice dynamics, with force constants from the measured phonon dispersions for $L1_2$ -ordered Cu_3Au .¹⁵ For the case of disordered Cu_3Au , a Born-von Kármán lattice dynamics model is not possible. For the disordered phase we used the same neutron-weight correction as for the ordered phase. Corrected phonon DOS curves of both ordered and disordered Cu_3Au are shown in Fig. 6. The DOS of the ordered alloy exhibits distinct van Hove singularities at 9, 13, 20, and 25 meV. The DOS of the disordered alloy has one very broad central feature around 16 meV, and two small peaks at 10 and 24 meV. The high-energy cutoff of the DOS is the same for the ordered and disordered alloys. Using these phonon DOS curves and Eq. (1), the phonon entropy of the disordered phase is larger than that of the ordered phase by $\Delta S_{\text{ph}}^{\text{d-o}} = 0.12k_{\text{B}}/\text{atom}$. This result is much smaller than that obtained with the virtual crystal approximation for the disordered alloy, which gave a value of $\Delta S_{\text{ph}}^{\text{d-o}} = 0.23k_{\text{B}}/\text{atom}$.¹⁹

Nagel *et al.*¹⁹ used the force constants of Katano *et al.*¹⁵ to evaluate the eigenvectors of the dynamical matrix of $L1_2$ Cu_3Au at the high symmetry points in the Brillouin zone. They identified the motions of atoms in various phonons, such as the highest frequency modes above 25 meV, which involved the motions of Cu atoms, primarily in the plane of their cube faces, with opposing movements of Au atoms. The Cu-Cu motions were found primarily in the modes around 20 meV. Nagel, *et al.* compared the dynamics

of the ordered alloy to that of the disordered alloy calculated with the risky virtual crystal approximation, and found that the modes around 20 meV in the ordered alloy undergo the greatest change with disorder. It was suggested that constraints on the motions of Cu atoms from the framework of the massive Au atoms, are a characteristic of the $L1_2$ structure of Cu_3Au . These constraints were assumed to be removed in the disordered alloy, causing a shift of the modes at 20 meV to lower energy.¹⁹ Although we do not have a clear picture of the lattice dynamics of the disordered alloys, the matched phonon DOS curves shown in Fig. 6 are consistent with this picture.

The difference of vibrational entropy between the ordered and disordered phases of Cu_3Au can also be obtained by using the neutron-weight corrected entropy of the disordered alloy, rather than the disordered DOS curve of Fig. 6. The phonon entropy of disordering depends sensitively on the phonon DOS of both the ordered and disordered phases of Cu_3Au . The DOS of ordered Cu_3Au is known more reliably than the DOS of the disordered phase, owing to the reliable neutron-weight correction of the ordered phase.²⁰ Most of the uncertainty in the value of $\Delta S_{\text{ph}}^{\text{d-o}}$ derived from the phonon DOS comes from the neutron-weight correction of the disordered phase. For the correction methods discussed in the previous sub-sections, the values of $\Delta S_{\text{ph}}^{\text{d-o}}$ can be obtained from the points in Fig. 4. Using an entropy of formation for the ordered alloy of $0.05 \pm 0.03k_{\text{B}}/\text{atom}$,²⁰ our best estimate of the phonon entropy of disordering in Cu_3Au is $0.15 \pm 0.05k_{\text{B}}/\text{atom}$ at 300 K. This result for $S_{\text{ph}}^{\text{d-o}}$ is consistent with the $0.14 \pm 0.05k_{\text{B}}/\text{atom}$ obtained by calorimetry,¹⁹ but somewhat larger than the $0.06k_{\text{B}}/\text{atom}$ calculated by Ozoliņš, Wolverton, and Zunger.⁸ The phonon entropy of disordering at T_c will be different, depending on how the phonon DOS of the two phases change between 300 and 663 K.

IV. CONCLUSIONS

Neutron-weighted phonon DOS curves were obtained from inelastic neutron scattering spectra measured on six disordered Cu-Au alloys at 300 K. The partial phonon entropies of Cu and Au are expected to vary smoothly as functions of composition. This allowed the development of neutron-weight corrections to the phonon entropy using models for the partial phonon entropies of Cu and Au atoms. We tested several schemes for obtaining a phonon entropy of alloying from a neutron-weighted phonon DOS curve. These phonon entropies of formation of disordered fcc Cu-Au alloys were generally in agreement with each other and with the results of a calculation by Ozoliņš, Wolverton, and Zunger using a local cluster approximation. We estimate a phonon entropy of disordering of $0.15 \pm 0.05k_{\text{B}}/\text{atom}$ for Cu_3Au at 300 K. A resonance mode associated with the motions of the heavy Au atoms in the Cu-rich alloys was observed at 9 meV. This resonance mode was within a reasonable agreement with predictions of mass-defect theory.

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