

# Equations of state of $\alpha$ , $\epsilon$ and liquid iron and iron's melting curve — Thermodynamic calculations

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**Abstract.** The melting curve between  $\epsilon$  and liquid iron ( $100\text{GPa} < P < 300\text{GPa}$ ) has been derived by computing Gibbs free energies at high pressures and high temperatures from equations of state of the  $\alpha$ ,  $\epsilon$  and liquid phases. The most uncertainty lies in the equation of state (EOS) of the  $\epsilon$  phase. By comparing the calculations to experimental data, the internal thermodynamic consistency of the three phases are examined. The best fits to the melting curves of *Boehler* [1993] and *Williams et al.* [1987] can be obtained with lower bulk moduli than determined by static compression. Using available equations of state of the iron phases, our calculations indicate that if sub-solidus iron is of the  $\epsilon$  phase, *Boehler's* melting curve is thermodynamically more consistent than *Williams et al.'s*. The problem is complicated by the possible existence of a new phase between the  $\epsilon$  and the liquid fields.

## Introduction

The melting curve under high pressures between  $\epsilon$  and liquid iron is directly related to the inner core-outer core temperature of the Earth. However, major disagreements exist between results from static and shock-wave measurements [*Boehler*, 1993; *Williams et al.*, 1987]. In this paper we attempt to determine the melting curve from equations of state of the  $\alpha$ ,  $\epsilon$  and liquid phases in a thermodynamically consistent manner. The benefit of this approach can be two-fold: (1) it may give preference to a certain melting curve; (2) it also highlights thermodynamic quantities which need to be better determined in order to further constraint the melting curve.

## Gibbs Free Energy Calculation Within a Single Phase

Two approaches we used in calculating Gibbs free energy ( $G$ ) of a single phase as a function of  $P$  and  $T$  are outlined below.

## Calculation Using an Isotherm-Isobar Mesh

In this approach [*Song and Ahrens*, 1994],  $G(P, T)$  is calculated from reference point  $G(P_0, T_0)$  by moving along the isobar to  $(P_0, T)$  then the isotherm to  $(P, T)$ :

$$G(P, T) = H(P_0, T) - TS(P_0, T) + \int_0^P V(T, P') dP' \quad (1)$$

where the first two terms are

$$H(P_0, T) = H(P_0, T_0) + \int_{T_0}^T C_p(P_0, T') dT' \quad (2)$$

$$S(P_0, T) = S(P_0, T_0) + \int_{T_0}^T \frac{C_p(P_0, T')}{T'} dT' \quad (3)$$

$H(P_0, T_0)$  and  $S(P_0, T_0)$  are enthalpy and entropy of formation from elements,  $V$  and  $C_p$  are volume and specific heat at constant pressure.

For the last term on the right hand side of Eq. 1,  $V(T, P')$  is again calculated on the  $P'$ - $T$  mesh: First,  $V(T_0, P')$  is calculated from, e.g., third order Birch-Murnaghan equation-of-state

$$P' = \frac{3}{2} K_T \left[ \frac{V(P_0, T_0)}{V(P', T_0)} \right]^{7/3} - \left[ \frac{V(P_0, T_0)}{V(P', T_0)} \right]^{5/3} \\ \times \left( 1 + \frac{3}{4} (K_T' - 4) \left[ \left( \frac{V(P_0, T_0)}{V(P', T_0)} \right)^{2/3} - 1 \right] \right) \quad (4)$$

given the reference volume, isothermal bulk modulus  $K_T$  and its derivative  $K_T'$ , then it undergoes thermal expansion to the final state:

$$V(P', T) = V(P', T_0) \exp \left[ \int_{T_0}^T \alpha(P', T') dT' \right] \quad (5)$$

The thermal expansion coefficient  $\alpha(P', T')$  is usually experimentally determined only at  $P' = 1\text{bar}$ . The pressure dependence assumed is:

$$\alpha(P', T') = \alpha(P_0, T') \left[ \frac{V(P', T')}{V(P_0, T')} \right]^\delta \quad (6)$$

where  $\delta$  is the second Grüneisen parameter which is taken to be constant.

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Gibbs energies of the  $\alpha$  and  $\epsilon$  phases of iron are calculated with this method.

### Calculation Using an Isentrope-Isometric Mesh

The calculation of the Gibbs energy of liquid iron employs:

$$G(P, T) = E(P, T) - TS(P, T) + PV(P, T) \quad (7)$$

At any  $P$  and  $T$ ,  $E$ ,  $S$  and  $V$  are calculated by first moving along an isentrope from  $(P_0, T_0)$  to  $(P_s, T_s)$ , at which the internal energy is  $E_s$  and volume is  $V_s$ :

$$E_s(P_s) = E(P_0, T_0) - \int_{V_0}^{V_s} P_s dV \quad (8)$$

$$T_s(P_s) = T_0 \exp \left[ - \int_{V_0}^{V_s} \left( \frac{\gamma}{V} \right) dV \right] \quad (9)$$

where  $V_s$  is again determined by third order Birch-Murnaghan EOS (same as Eq. 4 with isothermal bulk modulus and its pressure derivative substituted with their isentropic counterparts).  $\gamma$  is the Grüneisen parameter, for liquid iron, it has been fit as a function of internal energy, i.e., in Eq. 9,  $\gamma = \gamma(E_s)$ .

The second step is to move along an isometric to  $(P, T)$ . With  $V(P, T)$  held at constant  $V_s$ , either  $P$  or  $T$  are variable. We treat the final pressure  $P$  as a free parameter, therefore  $T = T(P, P_s)$ . The internal energy of the final state ( $E$ ) is given by solving the Mie-Grüneisen equation:

$$\int_{E_s}^E \gamma(E') dE' = V_s(P - P_s) \quad (10)$$

then the following equation is solved for  $T$ :

$$\int_{T_s}^T C_v(T', V_s) dT' = E - E_s \quad (11)$$

and  $S$  is

$$S = S_0 + \int_{T_s}^T \frac{C_v(T', V_s)}{T'} dT' \quad (12)$$

Eqs. 10–12 give all the thermodynamic quantities needed to evaluate Eq. 7, given values for  $K_s$ ,  $K'_s$ ,  $C_v$  and  $\gamma$ .

### Gibbs Energy of $\alpha$ and Liquid Iron

Huang *et al.* [1987] reported  $\alpha$  iron high-pressure compression data with the reference state at 1bar and 300K. We used Robie *et al.* [1979]'s data for  $C_p$ .

For liquid iron, the reference state is at the melting point (1809K) at 1bar.  $H_0$  and  $S_0$  are calculated from the values of the  $\alpha$  phase at the same pressure and temperature, and the enthalpy change upon melting. We used the EOS parameters of liquid iron in Anderson and Ahrens [1994a].

### Gibbs Energy of $\epsilon$ Iron

$K_T$  and  $K'_T$  for  $\epsilon$  iron have been measured by X-ray diffraction under static compression [Huang *et al.*, 1987; Mao *et al.*, 1990]. Thermal expansion coefficient is poorly constrained, previous estimates from static experiments exist in the 150–450°C, 10–20GPa region ( $\sim 3\text{--}5 \times 10^{-5} \text{K}^{-1}$  [Huang *et al.*, 1987]), shock wave data yield a mean value of  $\alpha$  from 300 to  $\sim 5200\text{K}$  at 202GPa ( $\sim 9 \times 10^{-6} \text{K}^{-1}$  [Duffy and Ahrens, 1993]). No experimental data is available for the specific heat. We calculated  $C_v$  by summing the contributions from the lattice (given by the Debye model) and from the electrons. For the electronic contribution ( $C_e$ ), we adopted the theoretical results of Boness *et al.* [1986]. We assumed the following value for  $C_v$ :

$$C_v(P, T) = c \left[ 9R \times D \left( \frac{T_D}{T} \right) + C_e \right] \quad (13)$$

where  $D$  is the Debye function, and Debye temperature for  $\epsilon$  iron is 385K [Andrews, 1973; Kerley, 1993].  $c$  is an *ad hoc* parameter: if our model is correct,  $c$  is unity.

$C_p$  is then calculated from  $C_v$  by

$$C_p = C_v + \alpha^2 V K_T T \quad (14)$$

The reference state is chosen to be (12GPa, 300K), at which the  $\alpha$  phase transfers to  $\epsilon$ . The entropy change of the transition  $\Delta S$  at 300K is unknown, but from the Clausius-Claperon relation:

$$\Delta S = \Delta V \frac{dP}{dT} \quad (15)$$

( $\Delta V$  is the volume change during the  $\alpha$ - $\gamma$  phase change at 300K [Huang *et al.*, 1987],  $\frac{dP}{dT}$  is the slope of the phase boundary in  $P$ - $T$  plane (reviewed in [Besson and Nicol, 1990]), it is estimated the enthalpy change  $\Delta H = T\Delta S$  is no more than  $\sim 0.6\text{kJ/mol}$ , or about 5% of the enthalpy of iron melting under 1 bar. In absence of more accurate data, we set  $\Delta H$  to zero in our calculations.

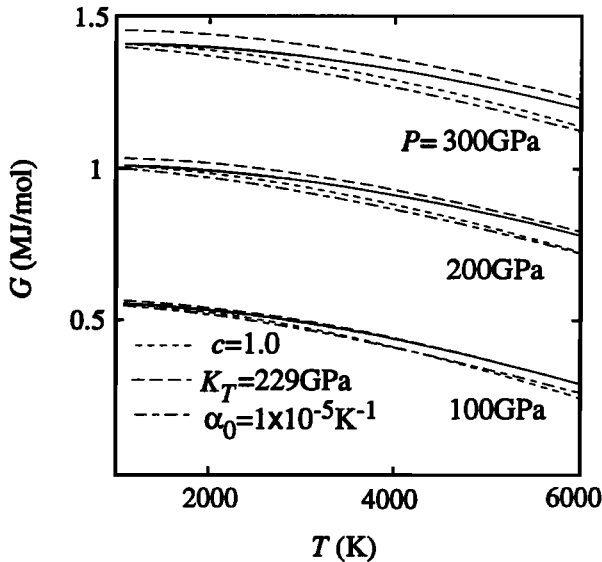
### Melting Curve Between the $\epsilon$ and Liquid Phases

After the Gibbs energies of the  $\epsilon$  and liquid phases are calculated independently, the melting curve is where the two Gibbs energy surfaces intersect. We keep the  $\alpha$  and liquid phases EOS fixed and adjust the  $\epsilon$  phase to fit experimentally determined melting curves [Boehler, 1993; Williams *et al.*, 1987].

In the following sections we discuss the effects of various parameters ( $C_p$ ,  $K_T$ ,  $\alpha$ ) of the  $\epsilon$  phase which are not well determined from experiments.

#### Specific Heat

We vary  $C_p$  by changing the parameter  $c$  in Eq. 13. It has a pronounced effect on the slope of the  $G_\epsilon(T)$  at a given pressure. The effect is nearly the same at different pressures (Figure 1(a)). At  $c=1$  (the theoretical value),



**Figure 1.** Effects of specific heat, bulk modulus and thermal expansion on  $G$ . Solid curve is for  $c = 0.9, K_T = 190\text{GPa}, K'_T=4, \alpha_0 = 4 \times 10^{-5}\text{K}^{-1}, \beta = 2 \times 10^{-5}$  and  $\delta = 3.2$ . Parameters changed for dashed curves are indicated.

$G_\epsilon(T)$  drops faster than  $G_l(T)$  at higher pressures (200–300GPa) such that they may never intersect. This is in clear contradiction with experimental data.

#### Bulk Modulus

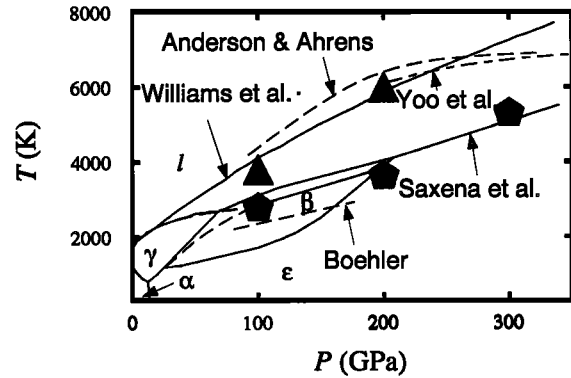
The bulk modulus controls the gradient of  $G_\epsilon(P)$  at a certain temperature (the spacing between 100, 200 and 300GPa curves in  $G$ - $T$  plane, Figure 1(b)). The effect decreases with  $T$  due to  $K_T$ 's temperature dependence, so  $K_T$  also effects the slope of  $G_\epsilon(T)$ , though much less directly than  $C_p$  does. The effect can further be fine-tuned by  $K'_T$ .

#### Thermal Expansion

$\alpha$  has a similar effect on  $G_\epsilon(P)$  at fixed pressures as  $C_p$  (smaller  $\alpha$  gives steeper  $G_\epsilon(P)$ , Figure 1(c)), but the change of slope has much stronger pressure dependence than that caused by  $C_p$ . Again, the effect can be subtly controlled with  $\alpha$ 's temperature dependence ( $\frac{1}{\alpha_0}(\frac{\partial \alpha}{\partial T})_p \equiv \beta$ ) and pressure dependence ( $\delta$ ).

### $P$ - $T$ Phase Diagrams of Iron

*Boehler* [1993] and *Saxena et al.* [1993] suggested a relatively low melting curve (although they disagree about the stability region of the new  $\beta$  phase, see Figure 2). *Saxena et al.* [1994] proposed the phase diagram shown in Figure 2 based on their experimental data and thermodynamic calculations similar to ours. *Williams et al.* [1987], *Yoo et al.* [1993] approximately agreed on a higher melting curve obtained completely or partially (data from *Bass et al.* [1987] are incorporated in the Williams paper) from shock wave experiments. Recently *Anderson and Ahrens* [1994b] revised shock temperature calculations of *Bass et al.* [1987] which arrived



**Figure 2.** Iron phase diagrams given by different groups. *Boehler's* and *Saxena's* phase diagrams are similar except the  $\beta$ - $\epsilon$  boundary. The melting curves from shock wave data are all higher than static curves. The solid-solid transition on the principal Hugoniot at  $\sim 200\text{GPa}$  observed by *Brown and McQueen* (not shown in graph) has a higher temperature than either *Boehler* or *Saxena's*  $\beta$ - $\epsilon$  transition temperature at that pressure. Filled symbols are numerical fits using EOS parameters in Table 1 for the  $\epsilon$  phase.

at a lower melting point at high pressures ( $> 230\text{GPa}$ ), but not enough to explain the difference between the static and shock wave data. The origin of the solid-solid transformation observed in the principal Hugoniot (at  $\sim 200\text{GPa}$  [*Brown and McQueen*, 1986]) before melting (at  $\sim 243\text{GPa}$ ) is uncertain. *Brown and McQueen* interpreted it as the  $\epsilon$ - $\gamma$  transition, while *Boehler* [1993] believed it is between  $\epsilon$  and  $\beta$ . *Anderson* [1994] offered a third scenario by suggesting yet another solid phase  $\theta$  above 200GPa and 4000K. Structure studies of the  $\beta$  phase are only theoretical. *Matsui* [1992] suggested it is bcc from his molecular simulation work, but *Stirzude and Cohen* [1994] concluded bcc structure is unstable in the  $\sim 150\text{GPa}$  range from density function theory. It is almost certain there is at least one phase between the  $\epsilon$  and liquid stability fields in the pressure range of 100–300GPa, but the equation of state of the phase is unknown and hence calculating its fusion curve is infeasible.

### Conclusions

Fits to *Boehler* [1993] and *Williams et al.* [1987]'s data are obtained (Figure 2) using two sets of parameters listed in Table 1. Notice the parameters are cen-

**Table 1.** EOS parameters used for  $\epsilon$ -iron to fit experimental melting curves

	$K_T(\text{GPa})$	$K'_T$
Boehler	205	4.8
Williams	120	7.5

Reference state is 12GPa and 300K. Other parameters (common to both fits) are  $c = 0.90, \alpha = 3 \times 10^{-5}\text{K}^{-1}, \beta = 2 \times 10^{-5}$  and  $\delta = 3.2$ .

tered at 12GPa, 300K. Extrapolated to ambient conditions,  $K_T^0$  (bulk modulus at 1bar and 300K) used to fit Boehler's data is 147GPa, significantly lower than the 204GPa (with  $K_T'=5$ ) reported by Huang *et al.* [1987]. The 147GPa value is closer to Mao *et al.* [1990]'s data (165GPa). Mao's data, however, yields a melting curve too low in the 100–300GPa range. Interestingly, the values of  $K_T^0 \approx 30$ GPa and  $K_T'=7.5$  are necessary to fit Williams *et al.*'s data. Therefore, assuming the phase diagram and the equations of state of the  $\alpha$  and liquid phases are approximately correct, our calculations favor Boehler's melting curve over Williams'. Finally, in either fit, the specific heat is 10% lower than its theoretical value.

Although we cannot completely delineate the effects of all variables on the Gibbs energy of the  $\epsilon$  phase, we note that to fit Williams' data, the bulk modulus must be lowered to fit the melting point at 100GPa ( $\sim 4000$ K). Although steepening the Gibbs energy temperature dependence (by, e.g., increasing the value of  $C_p$ ) would also raise the M. P. at 100GPa, it yields too high a M. P. at 200GPa. More experimental data on the  $\beta$  and  $\epsilon$  (especially its thermal expansion coefficient) would lead to more definite conclusions from our calculations.

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