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Field Effect Optoelectronic Modulation of Quantum-Confined Carriers in Black Phosphorus

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Abstract:

We report measurements of the infrared optical response of thin black phosphorus under field-effect modulation. We interpret the observed spectral changes as a combination of an ambipolar Burstein-Moss (BM) shift of the absorption edge due to band-filling under gate control, and a quantum confined Franz-Keldysh (QCFK) effect, phenomena which have been proposed theoretically to occur for black phosphorus under an applied electric field. Distinct optical responses are observed depending on the flake thickness and starting carrier concentration. Transmission extinction modulation amplitudes of more than two percent are observed, suggesting the potential for use of black phosphorus as an active material in mid-infrared optoelectronic modulator applications.

Keywords:

Black phosphorus, tunable optical properties, mid-infrared, Burstein-Moss shift, quantum-confined Franz-Keldysh effect, optical modulator

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The emergence of a variety of two-dimensional materials has spurred tremendous research activity in the field of optoelectronics¹⁻⁴. While gapless graphene can in principle exhibit an optoelectronic response at wavelengths ranging from the far infrared to the ultraviolet, its optoelectronic behavior is limited by a lack of resonant absorption and poor optical modulation in the absence of one-dimensional confinement. On the other hand, the semiconducting molybdenum- and tungsten-based transition metal dichalcogenides have shown considerable prospects for visible frequency optoelectronics. Yet while these materials promise exciting new directions for optoelectronics and nanophotonics in the visible range, they have limited response for lower energy, infrared light.

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The isolation of atomically thin black phosphorus in recent years has bridged the wavelength gap between graphene and transition metal dichalcogenides, as black phosphorus is an emerging two-dimensional semiconductor material with an infrared energy gap and typical carrier mobilities between those of graphene and transition metal dichalcogenides.⁵⁻⁹ Since the first isolation of black phosphorus and demonstration of a field effect device, numerous reports investigating the synthesis and optoelectronic properties of this material have emerged, appropriately summarized in recent reviews.^{5, 6, 10-12} Likewise a number of reports have also appeared on the applications of black phosphorus in fast photodetectors¹³, polarization sensitive detectors,¹⁴ waveguide integrated devices¹⁵, multispectral photodetectors¹⁶, visible to near-infrared absorbers¹⁷ and emitters,¹⁸⁻²¹ heterojunction²² and split gate p-n homojunction photovoltaics²³, gate-tunable van der Waals heterojunctions for digital logic circuits^{24, 25} and gigahertz frequency transistors in analog electronics²⁶. A majority of the studies on both the fundamental optical properties of black phosphorus and applications in optoelectronic devices have explored only the visible frequency range²⁷⁻³⁰. Therefore, little is known about the intrinsic optical response of black phosphorus in the infrared range. As a narrow band-gap semiconductor, much of the potential for black phosphorus lies in these infrared optoelectronic applications – ranging from tunable infrared emitters³¹ and absorbers for waste heat management/recovery³² to thermophotovoltaics³³ and optical modulators for telecommunications³⁴. Theoretical investigations of black phosphorus have suggested novel infrared optical phenomena, such as anisotropic plasmons^{35, 36},

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3 field-effect tunable exciton stark shifts³⁷, and strong Burstein-Moss³⁸ and quantum-
4 confined Franz-Keldysh effects³⁹ that promise to open new directions for both
5 fundamental nanophotonics research and applications. In this work, we report the first
6 experimental observations of the infrared optical response of ultrathin BP samples under
7 field effect modulation. We observe modulation of oscillations in the transmission spectra
8 which we attribute to a combination of an ambipolar Burstein-Moss shift and quantum-
9 confined Franz-Keldysh behavior.

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16 Measurements were performed on black phosphorous flakes that were
17 mechanically exfoliated in a glove box onto a 285 nm SiO₂/Si substrate. We analyzed
18 three flakes of 6.5 nm, 7 nm, and 14 nm thickness, determined by Atomic Force
19 Microscopy (details are provided in the Supporting Information Fig. S6), and lateral
20 dimensions of approximately 10 μm x 10 μm. A schematic of our experimental setup is
21 shown in Figure 1a. Standard electron beam lithography and metal deposition methods
22 were used to define Ni/Au electrodes to each exfoliated BP flake. The samples were then
23 immediately coated in 90nm PMMA for protection against environmental degradation.
24 Once encapsulated in PMMA we observe minimum degradation of our samples to
25 ambient exposure as verified by Raman spectroscopy⁴⁰ and reported in literature
26 precedent⁴¹. Transmission measurements were obtained via Fourier Transform Infrared
27 (FTIR) Spectroscopy. All optical measurements were done in a Linkam cryo-stage at a
28 pressure of 3 mTorr and a temperature of 80 K. First, a room-temperature gate-dependent
29 source-drain current was measured to extract approximate carrier densities as a function
30 of gate bias. Transmission spectra were then gathered at different gate voltages applied
31 between the flake and lightly doped Si substrate. We note that in our setup, the silicon
32 substrate is grounded and BP experiences the applied voltage, so the sign of the applied
33 voltages is reversed from the more common convention. In order to probe the electric
34 field- and charge-carrier-dependent optical properties of the BP, all spectra were
35 normalized to the zero-bias spectrum. The measured infrared optical properties result
36 primarily from the unique band structure of thin BP, schematically depicted in Figure 1b.
37 Quantized inter sub-band transitions provide the primary contribution to its zero-field
38 optical conductivity.

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We first present results for the 7 nm thick BP flake, in Figure 2. An optical image is shown in Figure 2e. FTIR spectra were taken using a Thermo Electron iS50 FTIR spectrometer and Continuum microscope for which the light source is a broadband, unpolarized tungsten glow-bar. To improve signal/noise and minimize spatial drift, we surrounded the sample with a 150 nm thick gold reflector which also served as the gate electrode. The extinction modulation results are presented in Figure 2a. We observe two major features in this flake at energies of 0.5 eV (I) and 0.9 eV (II). The dip in extinction at 0.5 eV is present for both positive and negative gate voltages, as the sample is increasingly hole or electron doped, respectively. It grows in strength as the doping is further increased at larger gate-biases. The same trend is true for the feature at 0.9 eV, where a smaller peak in extinction modulation is observed for both polarities of voltage. This peak also is strengthened as the gate voltage is increased to +/- 120V. To gain insights into this behavior, we measure gate-dependent transport, using a scheme in which a positive bias induces hole-doping, and a negative bias introduces electron-doping. We observe ambipolar transport at room temperature and atmospheric conditions, as shown in Figure 2b. Similar results have been shown in the literature with on/off ratios of $\sim 10^4$ for flakes thinner than the one considered here, at low temperature.^{22, 42} From this, the CNP is observed to be at 20 V, and, using the parallel plate model described in the Supporting Information, the unbiased, n-type carrier concentration is estimated to be $1.5 \cdot 10^{12} \text{ cm}^{-2}$. Further discussion on the depletion length and vertical charge distribution within the flake has been provided in the Supporting Information Figure S5.

We can interpret our spectroscopic results with consideration of a Burstein-Moss shift, which is a well-known phenomenon in chemically doped narrow-band gap semiconductor materials. This effect, which changes the optical band gap of a semiconductor, results from band-filling. As the charge carrier density is increased and the Fermi level moves into the conduction or valence band, there are fewer unoccupied electronic states available, and optical transitions to the occupied states are disallowed. This results in a decrease in the optical conductivity of the material at the energy of the transition, and is manifest in measurements as a decrease in absorption^{43, 44}. Because this flake exhibits ambipolar transport behavior, we can explain both features (I) and (II) as arising from an ambipolar BM effect. At zero applied bias, the flake is very lightly doped,

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3 and all optical transitions are allowed. As a positive gate voltage is applied and the
4 sample becomes hole doped, lower energy optical transitions become disallowed and the
5 absorption of the flake decreases. Feature (I) corresponds to the band filling effect of the
6 E_{11} intersubband transition, and feature (II) corresponds to the blocking of the E_{22}
7 intersubband transition, shown schematically in Figure 2d. For a negative gate voltage, as
8 the sample is electron-doped and the Fermi level moves into the conduction band, the E_{11}
9 and E_{22} transitions are again blocked due to band filling, resulting again in a decrease in
10 absorption. To support this explanation, we calculate the optical conductivity for the
11 flake, as shown in Figure 2c to identify the appropriate energies of the intersubband
12 transitions. To do so, we use the Kubo method described by Tony Low, et al.³⁸ The
13 observed transition energies are consistent with theoretical models that predict an
14 increase in band gap energy from the bulk 0.3 eV value as the material thickness
15 decreases to several layers or less.²⁹ This deviation from the bulk band gap indicates the
16 influence of vertical confinement of charge carriers, a feature of the two-dimensionality
17 of the material. We note that these transition energies suggest that the true thickness of
18 our sample is thinner than 7 nm, at approximately 4.5 nm. This apparent variation
19 between true and observed thickness from AFM topography is a result of surface
20 oxidation, as has been recently reported.⁴⁵ The surface oxide on our samples is expected
21 to be between 1-2 nm on either side, which appears inevitable despite following best
22 practices, and is stable with no measurable degradation over an ambient exposure of > 18
23 hrs in ambient (see Supporting Information). It is noteworthy that we observe extinction
24 modulation at relatively high photon energies, indicative of very large charge modulation
25 taking place in the fraction of the BP nearest to the silicon oxide interface, with an
26 accumulation/depletion layer that decays over the remainder of the flake. This is
27 consistent with in-depth calculations of charge screening in BP using the Thomas-Fermi
28 model done previously, reported by Tony Low, et al.³⁵ We estimate this screening length
29 to be of order 3 nm for our devices in the Supporting Information. This ambipolar, gate-
30 modulated Burstein-Moss shift is the first observed in a two-dimensional semiconductor,
31 to the best of our knowledge.

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55 We next present data for a BP flake of 14 nm thickness in Figure 3. An optical
56 image is shown in Figure 3e. Extinction measurements are again taken with an iS50 FTIR
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3 spectrometer and Continuum microscope for which the light source is a tungsten
4 glowbar. These results are presented in Figure 3a. Four prominent features are observed
5 to modulate under application of a gate voltage, at energies of 0.35 eV, 0.41 eV, 0.55 eV,
6 and 0.75 eV. As in the previous sample, they grow in strength with increased magnitude
7 of the gate voltage, regardless of polarity. To better understand this behavior, we again
8 measure gate-dependent transport, reported in Figure 4b. We observe ambipolar transport
9 characteristics as in the previous flake, centered about a conductance minimum at
10 approximately 20 V. Again using a parallel-plate capacitor model, we estimate an
11 unbiased n-type carrier density of $1.5 \cdot 10^{12} \text{ cm}^{-2}$ for a 20 V CNP.
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19 We propose that the optical modulation for this sample also results from an
20 ambipolar Burstein-Moss effect. In this case, as the Fermi energy is moved into the
21 conduction band of the BP under negative bias, transitions become disallowed and the
22 transmission is increased at each of the $E_{11} - E_{44}$ energies. Under positive bias, as the
23 Fermi energy is moved into the valence band, the band-filling effect of opposite charge
24 carrier type results in negative extinction modulation peaks at the same energies of
25 transitions $E_{11} - E_{44}$. As in the previous sample, we estimate an oxide layer of 1-2 nm has
26 grown on our BP on either surface. Based on optical conductivity calculations presented
27 in Figure 3c, we again estimate the adjusted thickness of our flake to be less than that
28 measured by AFM, at approximately 10 nm. We further note that for this sample, the
29 measurement extended beyond the area of the flake, to cover the flake and an area of bare
30 silicon oxide roughly eight times the flake are. We thus suggest that the true modulation
31 strength of this device is of order six percent, not the 0.75 percent indicated by the
32 modulation of the entire area.
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44 Finally, results for the 6.5 nm thick flake are reported in Figure 4, for which an
45 optical image is shown in Figure 4e. Unlike the previous two flakes, transmission
46 measurements for this sample were taken using a Nicolet Magna 760 FTIR spectrometer
47 coupled to a Nic-Plan infrared microscope on infrared Beamline 1.4.3 at the Advanced
48 Light Source (ALS) at Lawrence Berkeley National Laboratory. This allowed us to
49 perform measurements using a high brightness, diffraction-limited infrared beam, which
50 is beneficial for accurately analyzing the small-area BP samples attainable by mechanical
51 exfoliation. In contrast to the previous measurements, the incident light was elliptically
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3 polarized due to the synchrotron source, with an intensity ratio of two to one. The major
4 axis and details of the polarization state are indicated and discussed in Supporting
5 Information Figure S3.
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9 Figure 4a shows the primary result of this experiment, which is the modulated
10 extinction of the sample at different voltages, normalized to the zero-bias extinction
11 spectrum. Three prominent features are observed in these spectra. First, under negative
12 applied bias (i.e.: when the sample is being depleted of holes), a negative peak (I) appears
13 in transmission near 0.45 eV, which grows in amplitude and broadens to lower energies
14 as the magnitude of the bias increases. Second, under positive applied bias (i.e.: when
15 the sample is being increasingly hole-doped), a positive peak (II) appears in transmittance
16 near 0.5-0.7 eV. Lastly, these two effects, which we propose to depend on the Fermi
17 level, are superimposed with an oscillatory feature (III) that varies with the magnitude of
18 the applied field, but not its polarity, and which is most clearly visible in the negative bias
19 spectra in the 0.5 - 0.7 eV range.
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28 To better understand these results, transport measurements were again taken at
29 room temperature under ambient conditions, as shown in Figure 4b. The gate dependence
30 of the conductance indicates that, unlike the previous samples, this BP flake was initially
31 heavily hole-doped, as ambipolar transport is not observed and only hole-type conduction
32 is seen even at large negative bias.
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37 Due to the distinct character of each feature and their relation to the transport
38 measurements, we can understand the overall spectral shifts as arising from a
39 combination of a Burstein-Moss (BM) shift and a quantum confined Franz-Keldysh
40 (QCFK) effect, both of which have been predicted theoretically for gated BP flakes of
41 this thickness.³⁹ In the bulk limit, the Franz-Keldysh effect refers to electron and hole
42 wavefunctions leaking into the band gap, as described by Airy functions. This behavior
43 introduces oscillatory features to the interband absorption spectrum, and redshifts the
44 band edge. In confined systems, the quantum-confined Franz-Keldysh effect similarly
45 modulates intersubband transitions.⁴⁶ As confinement becomes stronger and excitonic
46 effects dominate, this phenomenon eventually gives way to the quantum-confined Stark
47 effect. Because our flake exceeds a thickness of ~ 4 nm, we expect excitonic effects to be
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weak and therefore will not focus our discussion on the quantum-confined Stark effect or a normal-to-topological phase transition in our analysis.^{29, 30, 37}

We suggest that peak (I) at 0.45 eV can be described by the onset of $j = 1$ intersubband transitions as the material is depleted of holes at negative gate voltages and the valence band is un-filled, in agreement with our transport measurements. We further suggest that peak (II) can be described primarily by the suppression of $j = 2$ inter-subband transitions as more holes are accumulated in the flake at positive gate voltages. This behavior is shown schematically in Figure 4d, and is again supported by calculations of the optical conductivity of the flake for various doping levels, shown in Figure 4e. Our experimental results correspond to modulation of the calculated intersubband transitions only in part, suggesting that a simple Burstein-Moss shift is insufficient to explain this measurement. From these results, we assign the band gap energy of our flake to be approximately 0.4 eV. Unlike our previous samples, the optical data indicates minimal oxide formation, as the E_{11} and E_{22} transition energies match well to theory for a 6.5 nm thick BP quantum well. Given we do not see the charge neutral point in transport, we do not assign a carrier density to this flake, but can say that with a charge neutral point of greater than -80 V, its p-type carrier density must be greater than $6 \cdot 10^{12} \text{ cm}^{-2}$.

We suggest that quantum-confined Franz-Keldysh effects lead to the appearance of the additional oscillatory spectral features we observe. Specifically, we point to the oscillations in the negative voltage extinction curves at energies above 0.5 eV – where Burstein-Moss considerations would predict zero modulation – and in the positive voltage extinction curves both in that same range – where Burstein-Moss behavior would predict only a single dip in extinction centered at the 0.575 transition energy – and at 0.45 eV. This oscillatory modulation increases with bias magnitude, but does not depend significantly on the sign of the bias -- behavior which is consistent with shifting of the overlap of the first and second conduction and valence sub-band wavefunctions, as described by the quantum-confined Franz-Keldysh effect. This behavior is investigated theoretically for gated BP by Charles Lin, et al.³⁹ In addition, under a sufficiently strong electric field, hybrid optical transitions between sub-bands of different index (eg: E_{v1} to E_{c2}) that are nominally forbidden at zero field become allowed. In total, quantum-confined Franz-Keldysh effects in thin BP are expected to lead to behavior including

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3 redshifting of intersubband transitions, modification of intersubband selection rules
4 (allowing hybrid transitions), or oscillatory, Airy function modulation of the absorption
5 edge, all of which can be considered as consistent with our experimental observations.
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7 However, further theoretical work is needed to understand this effect satisfactorily; the
8 same authors provide evidence in a more recent, experimental report that hybrid
9 transitions may occur with zero applied field as well.⁴⁷ Interestingly, we see no evidence
10 of a tunable plasma edge; investigations in the long-wave infrared wavelength range with
11 larger samples would likely be needed to observe this feature.
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18 The clear appearance of the QCFK effect in this measurement distinctly differs
19 from our previous two samples, indicating that BP quantum wells of similar thickness
20 may have very different optical responses. We suggest that the primary reason for this is
21 that this flake is very heavily doped under zero bias, whereas our previous measurements
22 were performed on nearly intrinsic flakes. In particular, in the intrinsic case, field strength
23 and carrier concentration vary proportionally (ie: under larger bias, there is a larger
24 carrier concentration, and vice-versa). To the contrary, in our heavily doped sample, this
25 proportionality is absent, leading to potentially competing effects and the clear
26 emergence of oscillatory features. It is also worth noting that, while we see no clear
27 evidence of the QCFK effect in our first two experiments, it is possible that the large BM
28 shift is simply dominant over the QCFK effect, making the latter effect difficult to
29 observe, or that our increased noise prevents the effect from obviously manifesting. A
30 complete theoretical framework that addresses the interplay between zero-bias carrier
31 concentration and field-effect has not yet been developed, and is beyond the scope of this
32 paper. We also note that, while we see no clear evidence of excitonic effects, and it has
33 been suggested theoretically and experimentally that such effects should not be present in
34 flakes of this thickness, we do not rule out the possibility that they may be influencing
35 our results.
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49 We note that because of the complicated polarization state of incident light from
50 the synchrotron, and because a previous study has extensively studied this effect
51 experimentally⁴⁸, we do not address in detail the anisotropic optical properties of BP.
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53 However, due to the primary contribution to the optical conductivity arising from the σ_{xx}
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3 component, we argue that the only effect of elliptically polarized light is to scale the
4 observed modulation, as discussed in the Supplement Section 1 - 3.
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7 In conclusion, we have demonstrated experimentally that ultra-thin black
8 phosphorus exhibits widely tunable, quantum well-like optical properties at mid-infrared
9 wavelengths. In 7 and 14 nm, lightly doped flakes, we observe for the first time an
10 ambipolar Burstein-Moss shift of intersubband transitions, which also varies with
11 thickness as these transition energies are changed. In a heavily doped 6.5 nm thick BP
12 flake, modulation of infrared transmission takes place as a result of both a Burstein-Moss
13 shift and additional, quantum-confined Franz-Keldysh effects. While our results verify
14 some of the recent theoretical predictions about the electro-optical effects in few-layer
15 BP, they also report new behavior and serve as motivation to further understand the BP
16 optical response as function of sample thickness, doping and field. Our results indicate
17 that BP is both an interesting system for exploring the fundamental behavior of quantum-
18 confined carriers in two-dimensional semiconductors under field-effect modulation, and a
19 promising candidate for tunable mid-infrared optical devices.
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31 **Methods:**

32 BP flakes were exfoliated in a glove box from crystals grown by HQ Graphene. After
33 fabrication of Ni/Au (20 nm / 130 nm) electrodes by electron beam lithography and
34 electron beam evaporation, 90 nm PMMA 950 A2 was spin-coated as an encapsulation
35 layer. Electron beam lithography was again used to expose the contacts for wire bonding.
36 PMMA¹⁰ and other encapsulation layers including ALD grown dielectrics,^{49, 50}
37 polymers^{41, 51}, covalent surface functionalization⁵² and atomically thin hexagonal boron
38 nitride⁵³ have been shown in the past to successfully protect BP devices against ambient
39 degradation.
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49 **Supporting Information:**

50 Section I shows extinction modulation of a 25 nm thick BP flake. Section II presents
51 methods for determining the crystal orientation of the samples. Section III discusses the
52 polarization state of the synchrotron beam used in some measurements and its anticipated
53 effects. Section IV gives the model used to estimate carrier concentration in our flakes.
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3 Section V presents our model for determining the accumulation/depletion length in BP
4 flakes. Section VI presents Atomic Force Microscopy measurements for thickness
5 determination of flakes. Section VII discusses the techniques used to prevent sample
6 oxidation and includes Raman measurements confirming minimal degradation.
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10 11 **Acknowledgments:**

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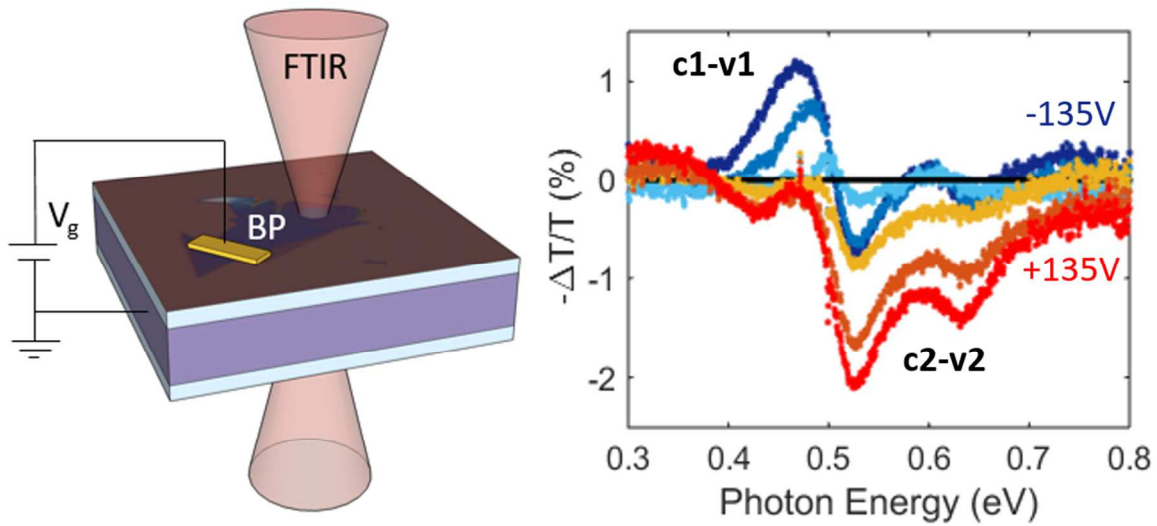


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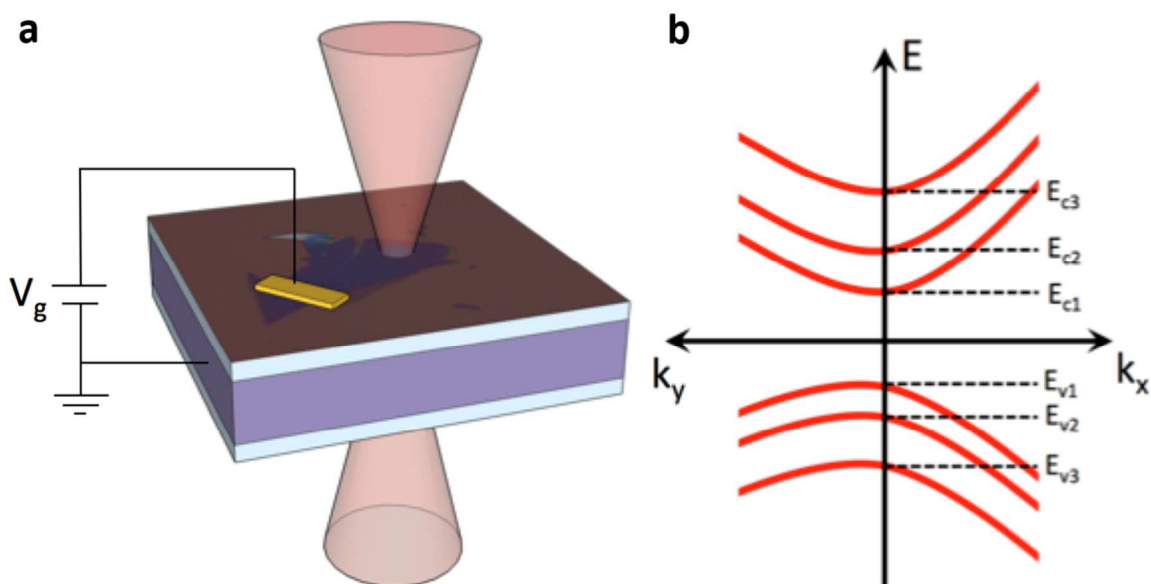


Figure 1: **a)** Schematic illustration of transmission modulation experiment. Broadband mid-IR beam is transmitted through black phosphorus sample. Variable gate voltage applied across SiO₂ modulates transmission extinction, **b)** Schematic band diagram of few-layer black phosphorus with subbands arising from vertical confinement

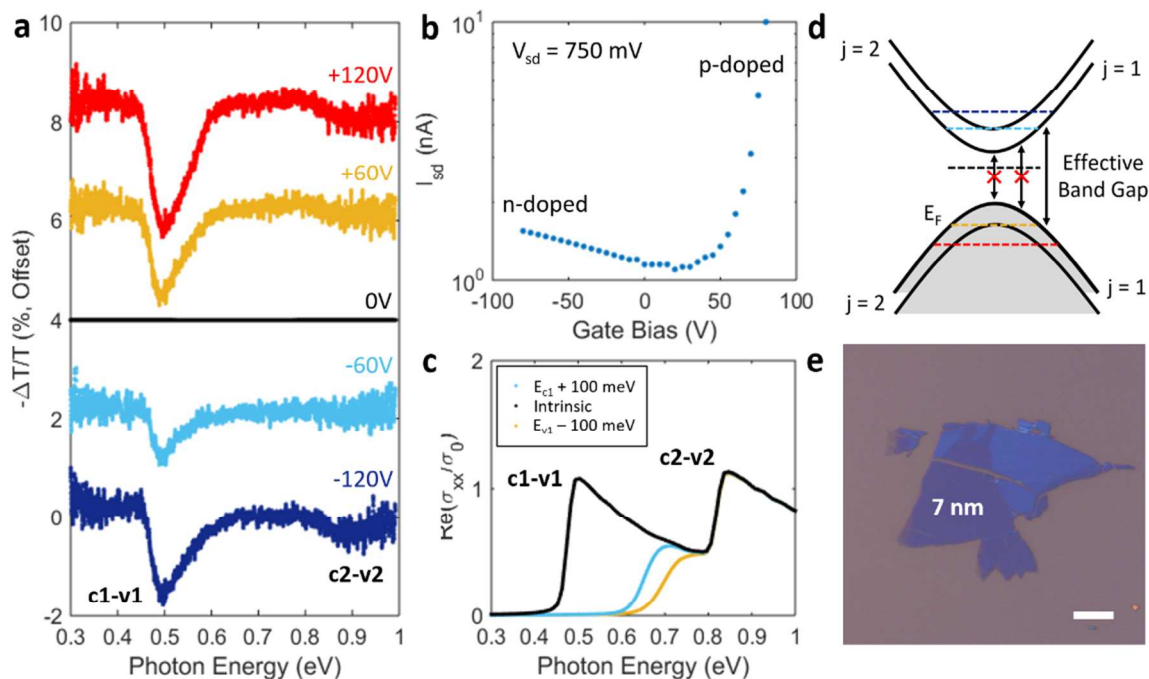


Figure 2: Gate modulation of lightly doped 7 nm flake. **a)** FTIR transmission extinction vs photon energy normalized to zero bias **b)** Source-drain current vs gate voltage. Ambipolar conduction is seen. **c)** Calculated optical conductivity of a 4.5 nm thick BP flake at different carrier concentrations, normalized to the universal conductivity of graphene. No field effects included. **d)** Schematic of electronic band structure and allowed interband transitions at different voltages. **e)** Optical microscope image of flake. Scale bar is 10 μ m.

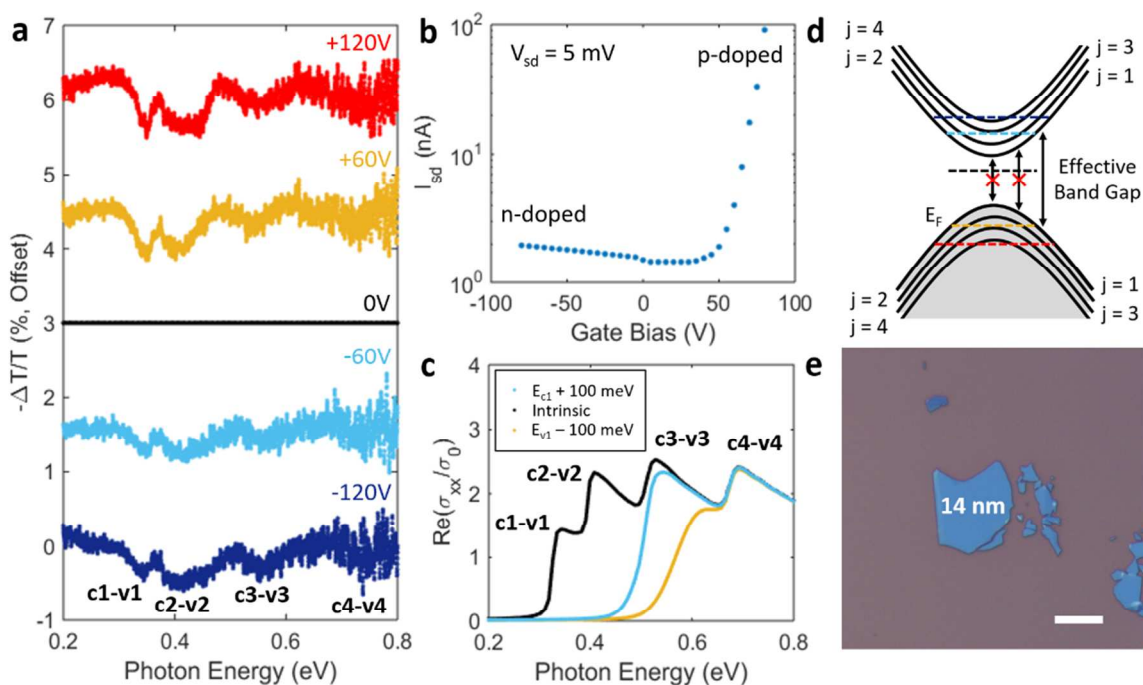


Figure 3: Gate modulation of lightly doped 14 nm flake. **a)** FTIR transmission extinction vs photon energy normalized to zero bias **b)** Source-drain current vs gate voltage. Ambipolar conduction is seen. **c)** Calculated optical conductivity of a 10 nm thick BP flake at different carrier concentrations, normalized to the universal conductivity of graphene. No field effects included. **d)** Schematic of electronic band structure and allowed interband transitions at different voltages. **e)** Optical microscope image of flake. Scale bar is 10 μm .

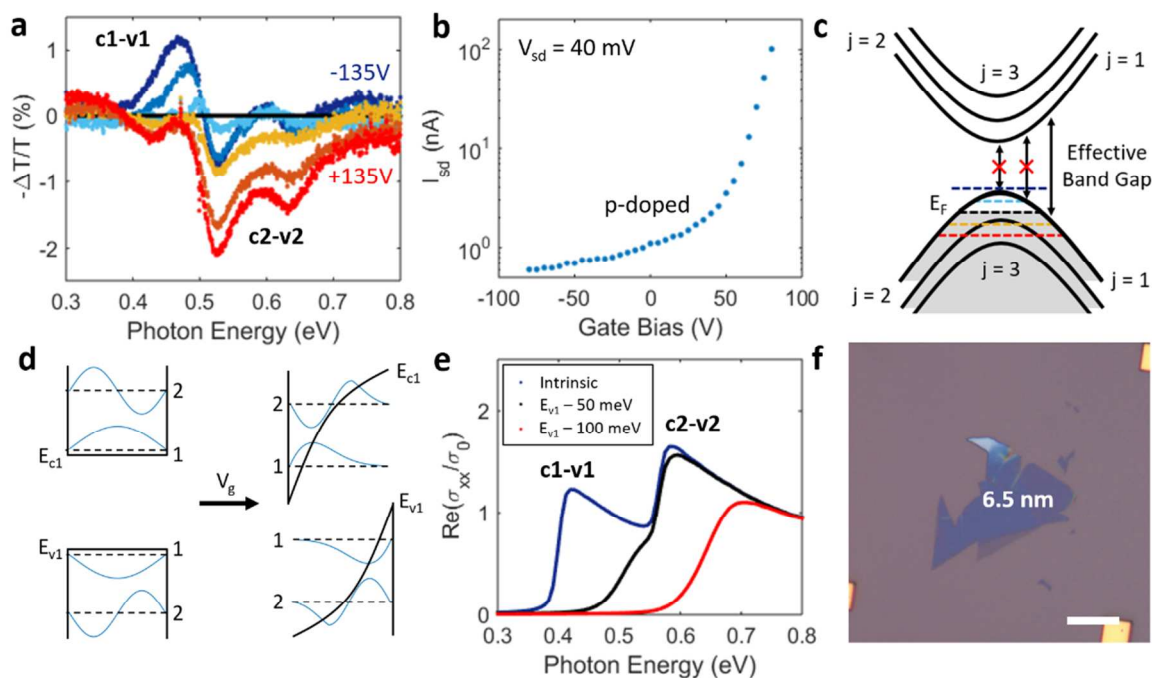


Figure 4: Gate modulation of a heavily doped 6.5 nm flake. **a)** FTIR transmission extinction vs photon energy normalized to zero bias **b)** Source-drain current vs gate voltage. Only hole-type conduction is seen. **c)** Schematic of electronic band structure and allowed interband transitions at different voltages. **d)** Schematic representation of quantum confined Franz-Keldysh Effect **e)** Calculated optical conductivity of a 6.5 nm thick BP flake at different carrier concentrations, normalized to the universal conductivity of graphene. No field effects included **f)** Optical microscope image of flake. Scale bar is 10 μm .