



# Synergy between Fe and Ni in the optimal performance of (Ni,Fe)OOH catalysts for the oxygen evolution reaction

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Contributed by William A. Goddard III, April 14, 2018 (sent for review December 18, 2017; reviewed by Daniel G. Nocera and Annabella Selloni)

The oxygen evolution reaction (OER) is critical to solar production of fuels, but the reaction mechanism underlying the performance for a best OER catalyst, Fe-doped NiOOH [(Ni,Fe)OOH], remains highly controversial. We used grand canonical quantum mechanics to predict the OER mechanisms including kinetics and thus overpotentials as a function of Fe content in (Ni,Fe)OOH catalysts. We find that density functional theory (DFT) without exact exchange predicts that addition of Fe does not reduce the overpotential much. However, DFT with exact exchange predicts dramatic improvement in performance for (Ni,Fe)OOH, leading to an overpotential of 0.42 V and a Tafel slope of 23 mV/decade (dec), in good agreement with experiments, 0.3–0.4 V and 30 mV/dec. We reveal that the high spin  $d^4$  Fe(IV) leads to efficient formation of an active O radical intermediate, while the closed shell  $d^6$  Ni(IV) catalyzes the subsequent O–O coupling, and thus it is the synergy between Fe and Ni that delivers the optimal performance for OER.

density functional theory | B3PW91 | PBE | electrocatalysis | reaction mechanism

**A**rtificial photosynthesis (AP) using solar energy to convert  $\text{CO}_2$  and  $\text{H}_2\text{O}$  into fuels and  $\text{O}_2$  is a most promising approach to a carbon-neutral cycle and scalable energy storage (1, 2). Electrocatalysis provides an attractive candidate route to AP, which could extend to all intermittent renewable energy resources. Another key advantage is the decomposition of AP reaction into two electrochemical half-reactions,  $\text{CO}_2$  reduction reaction ( $\text{CO}_2\text{RR}$ ) and oxygen evolution reaction (OER), that can use drastically different catalysts for optimal performance.

The rational design toward ideal electrocatalysts requires atomistic understanding of the reaction mechanisms for the target electrochemical reactions. Here, modern quantum mechanics (QM) methods can play an essential role. Indeed, the atomistic mechanisms underlying  $\text{CO}_2\text{RR}$  catalyzed by Cu and Cu-derived electrodes have been elucidated recently by using new methods of QM, including grand canonical (GC; constant electrochemical potential) QM and QM-based free energy sampling by metadynamics with full solvent, providing guidelines for efficient and selective catalyst design (3–7). However, such QM calculations of the mechanisms for OER, including reaction barriers and accurate solvent corrections, have only been reported for  $\text{IrO}_2$  (110) (8), which is complicated by difficulties in common density functional theory (DFT) methods in describing the spin triplet  $\text{O}_2$  product.

In this work, we examine the OER mechanism for a most efficient electrocatalyst under alkaline conditions (9), the Fe-doped NiOOH [(Ni,Fe)OOH], which remains elusive, and even the catalytic active site (Ni or Fe) is under debate (10–12). Here, we apply a state-of-the-art theoretical method, using the hybrid functional B3PW91 (13) and explicit constant electrochemical potential ( $\mu_e$ ) calculations (14, 15), to determine the atomistic mechanisms for OER on pure and Fe-doped NiOOH, including the kinetics of O–O coupling and  $\text{O}_2$  release. We find that

the efficiency of OER is dictated by two elementary steps, the formation of an active O radical species and the subsequent O–O coupling. However, the pure Ni system is inefficient for generating the O radical. We find that in the (Ni,Fe)OOH systems, the surface high spin  $d^4$  Fe(IV) site delivers the O radical more easily, while it is still the surface closed shell  $d^6$  Ni(IV) site that better catalyzes the O–O coupling. Thus, it is the synergy between Fe and Ni that offers the optimal performance of (Ni,Fe)OOH for catalyzing OER. This implies promising directions for further optimization toward ideal OER catalyst.

## Results and Discussion

**Composing the Appropriate Methodology.** Previous theoretical efforts to model the mechanism for OER on (Ni,Fe)OOH suffer several inadequacies. First, the Ni/Fe oxide/hydroxide systems feature localized unpaired spins that demand theory beyond the standard density functional approximations [local density approximation and generalized gradient approximation (GGA)]. A commonly used strategy for remedying this is to introduce the on-site Coulomb interaction (+U) (10, 16–18), but it is questionable for modeling chemistry, because the U term generally varies as functions of the electronic configurations of transition metal (TM) atoms (19, 20). Indeed, GGA(PBE)+U has been shown to be unreliable for predicting redox potentials of Ni/Fe oxyhydroxides (12).

A more general approach is including the Hartree–Fock exchange as in the hybrid functional methods to describe the localized electrons self-consistently (13, 21), and a recent study

## Significance

The oxygen evolution reaction (OER) is a key component to applications of electrochemistry in energy and environment, but the reaction mechanism for OER on a best-performing catalyst, the Fe-doped NiOOH [(Ni,Fe)OOH], is under debate. Using advanced quantum mechanics (QM) methods, we find that the (Ni,Fe)OOH systems feature synergy between Fe and Ni, in which the two components assume the two different key functionalities needed for OER, forming an active O radical and catalyzing the O–O coupling, respectively. It is this synergy that delivers the optimal performance of (Ni,Fe)OOH for driving the OER. This suggests a strategy to further improve the (Ni,Fe)OOH-type OER catalysts, i.e., separately screening for the best component for each key functionality.

Author contributions: H.X. and W.A.G. designed research; H.X. and H.S. performed research; H.X. and H.S. analyzed data; and H.X. and W.A.G. wrote the paper.

Reviewers: D.G.N., Harvard University; and A.S., Princeton University.

The authors declare no conflict of interest.

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This article contains supporting information online at [www.pnas.org/lookup/suppl/doi:10.1073/pnas.1722034115/-DCSupplemental](http://www.pnas.org/lookup/suppl/doi:10.1073/pnas.1722034115/-DCSupplemental).

Published online May 21, 2018.

of (Ni,Fe)OOH did use one of the flavors (PBE0), but focused on only the bulk phases (12).

Second, previous theoretical studies examined only the thermodynamics of intermediates for OER on (Ni,Fe)OOH, with no consideration of reaction kinetics [transition states (TSs) and associated barriers] (10, 18). Such disregard could be reasonable for the OER steps involving simple deprotonation processes associated with electrochemical oxidation, which are probably fast with barriers of a few kilocalories per mole as the proton transfer in aqueous electrolytes. But the key catalytic step before the final  $O_2$  production is O–O coupling that is likely to have a significant barrier. Indeed, our recent study of O–O coupling during OER on  $IrO_2$  found moderate barriers of 0.5–0.6 eV (8). Thus, to investigate the reaction kinetics for OER reactions, we can expect that exact exchange is required for predicting the kinetics accurately enough to predict reliable current densities and hence overpotentials for comparison with experiments.

Another important aspect of our calculations compared with previous theoretical modeling is our implementation of GC DFT (15) using the accurate CANDLE implicit electrolyte model (14). This allows us to describe the kinetic processes as a function of explicit  $\mu_e$  in describing electrochemistry at the interface; this success was previously demonstrated for  $CO_2RR$  on Cu(111) (3, 4, 7) and OER on  $IrO_2(110)$  (8). This is in contrast to the standard QM methods in which the number of electrons is kept fixed during the reactions, leading necessarily to changes in the applied potential ( $U$ ).

Here, we calculate the relative free energies by combining several components: (i) the internal energies from the hybrid functional B3PW91 calculations using the PBE optimized structures; (ii) the zero-point energies, enthalpy, and entropy contributions from PBE calculations; (iii) the solvation energies; and (iv) the explicit dependence on the  $U$  ( $U$  is referenced to the standard hydrogen electrode) calculated with PBE. *SI Appendix, SI Methods* provides the details. Note that the free energies and barriers mentioned below depend on  $U$ .

**The Structural Model for the Active (Ni,Fe)OOH Catalyst.** The OER active phase of (Ni,Fe)OOH systems has been identified as  $\gamma$ -NiOOH-like with a mixed formal oxidation state (FOS) of +3.6 for TM elements (10, 12, 22). Thus, to model the OER active phase, we use a model crystal structure (Fig. 1A) proposed by Ceder and coworkers based on theoretical analysis (23). This model consists of 2D  $(Ni,Fe)O_2$  layers intercalating with  $H_2O$  molecules and  $K^+$  ions that result in an interplanar spacing

of 6.8 Å. This matches well the current knowledge of  $\gamma$ -phase TM oxyhydroxides (9). The model structure assumes the chemical formula of  $K_{1/3}(H_2O)_{2/3}(Ni,Fe)O_2$ , which gives the proper average FOS of +3.67 for Ni/Fe.

Using the magnetic moment analysis suggested by Hammes-Schiffer and coworkers to determine the FOS of each TM atom (12), our DFT calculations at the B3PW91 level predict that, for the pure Ni case, the Ni atoms closest to  $K^+$  are of +3 FOS with a low spin  $d^7$  configuration, while the remaining 2/3 of the Ni atoms are of +4 FOS with a closed shell  $d^6$  configuration (Fig. 1A).

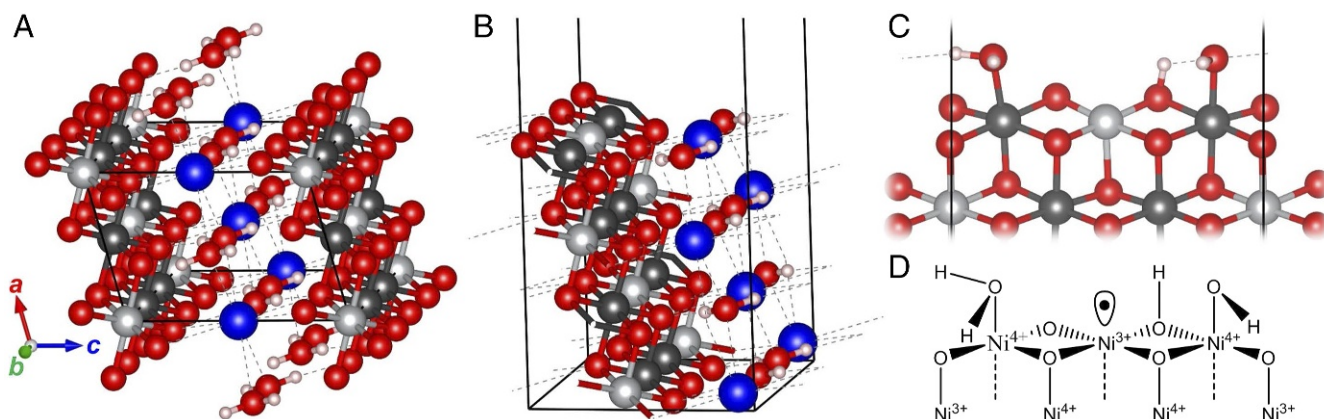
Introducing the Fe dopant, we find that, it preferably substitutes at the  $Ni^{4+}$  site as  $Fe^{4+}$  with a high spin  $d^4$  configuration. This is consistent with previous experimental (Stahl and coworkers) (24) and theoretical (Hammes-Schiffer and coworkers) (12) confirmation that  $Fe^{4+}$  is present as the onset of OER activity. *SI Appendix, SI Additional Discussions* provides additional discussion of Fe doping in NiOOH.

We consider the (100) surface of the adopted model structure that exposes the nonpolar edge of the  $(Ni,Fe)O_2$  layer (Fig. 1B), which has been shown by Bell and coworkers to be the highly OER active surface in similar systems (25, 26). This surface has three TM sites per layer, of which two are of +4 FOS and the other one is of +3 FOS. Letting the pristine surface of the pure Ni case come into contact with the electrolyte, we predict that the two  $Ni^{4+}$  sites react with  $H_2O$  ( $\rightarrow H_2O$ ) molecules chemically with one dissociated over the bridging O ( $O_b$ ) and neighboring  $Ni^{4+}$  site, while the  $Ni^{3+}$  site repels the  $H_2O$  molecule due to the unpaired electron. This is shown in Fig. 1C and D.

As described in *Composing the Appropriate Methodology*, we use the CANDLE implicit model (14) to describe the electrolyte. Thus, we introduce explicit  $H_2O$  molecules only when they chemically bind to the surface or become directly involved in the OER (The details of these structures are provided in *SI Appendix, SI Structural Information: Coordinates for All Predicted Intermediates and Transition States*).

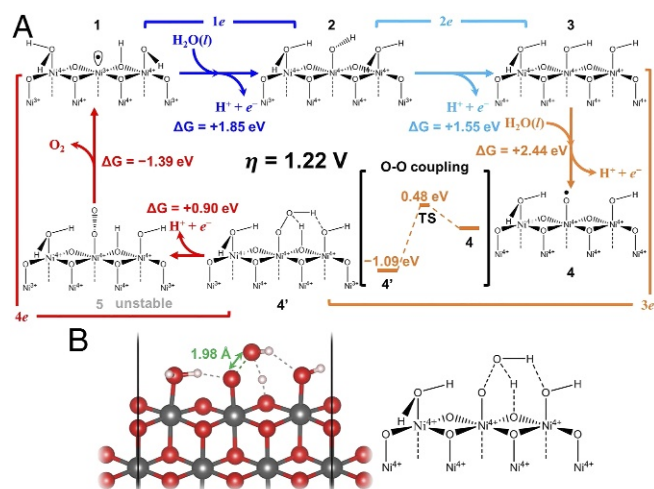
**The Reaction Pathway for OER on the Pure Ni System.** To determine the reaction pathway for OER on the pure Ni system, we examine every possible option for deprotonation associated with each electrochemical oxidation step (losing one electron) to find the lowest free energy state.

- a. The four electron steps for the pure Ni case (Fig. 2A)
  0. We start with state 1 with 2/3  $H_2O$  coverage,



**Fig. 1.** Structural models for the OER active (Ni,Fe)OOH catalysts. (A) The model crystal structure for the OER active  $\gamma$ -phase of (Ni,Fe)OOH catalysts, with chemical formula  $K_{1/3}(H_2O)_{2/3}(Ni,Fe)O_2$  [light gray for  $Ni^{3+}$ , dark gray for  $(Ni,Fe)^{4+}$ , blue for  $K^+$ , red for O, and white for H]. (B) The highly OER active (100) surface of A. (C) The configuration of water chemisorption on the active surface of the pure Ni system. (D) The chemical illustration corresponding to C. Note that the intercalating  $K^+$  and  $H_2O$  are omitted for clarity in C and D.





**Fig. 2.** Mechanism for OER on pure  $\gamma$ -NiOOH catalyst. (A) The mechanistic cycle of OER catalyzed by pure  $\gamma$ -NiOOH surface [B3PW91 predicted free energy differences  $\Delta G$  and barrier at  $U = 2.28$  V ( $\eta = 1.22$  V) corresponding to  $j = 10$  mA/cm<sup>2</sup>]. (B) The structure of TS for O–O coupling.

1. then the first electron ( $1e$ ) step features deprotonating the  $O_b$  while bringing in a third  $H_2O$  to hydroxylate  $Ni^{4+}$  that is oxidized from the  $Ni^{3+}$  site. This results in state 2 with 1/3 OH coverage and 2/3  $H_2O$  coverage.
2. Instead of further deprotonating the adsorbed OH ( $^*OH$ ) to a presumably active O species, we find that the second electron ( $2e$ ) step favors deprotonating one  $^*H_2O$  to produce state 3 with 2/3 OH coverage.
3. It is the third electron ( $3e$ ) step that generates a stable O species as in state 4, requiring a large free energy input of  $\sim 2.4$  eV. This O species is predicted to be a radical by both PBE and B3PW91 calculations, which give spin populations on O of 0.6 and 1.0, respectively. Thus, we identify by theory that the nature of the active O species is an O radical, consistent with previous experiments (22, 27–31). Consequently, we find that the closed shell  $d^0$   $Ni^{4+}$  is ineffective in producing this key active intermediate. We find that the alternative pathway through  $Ni(III)$ -O $\bullet$  is unfavorable (SI Appendix, SI Additional Discussions).

Interestingly, state 4 is not the most stable at the  $3e$  step, but when a fourth  $H_2O$  is introduced, we find that state 4' with an adsorbed OOH ( $^*OOH$ ) and hydrogenated  $O_b$  is lower in free energy by 0.7–1.1 eV. Our B3PW91 calculations predict a moderate free energy barrier of 0.5–0.6 eV going from state 4 to 4'. This TS features simultaneous O–O bond forming and protonating  $O_b$  by the incoming  $H_2O$  (Fig. 2B). The O–O coupling within the  $3e$  step does not involve losing any electron and thus is not an electrochemical step but a chemical one.

We note that there have been two prevailing mechanisms for O–O coupling in OER on TM oxo complexes, the water hydrogen atom abstraction (WHAA) mechanism and the intramolecular oxygen coupling (IMOC) mechanism (31, 32). Our mechanism resembles the WHAA mechanism, but differs significantly in the aspect that the WHAA mechanism involves two O radicals for O–O coupling and H abstraction from  $H_2O$ , respectively, but our mechanism involves only one O radical, and the  $O_b$  abstracts the H atom from  $H_2O$ .

4. After the state 4' is formed, the fourth electron ( $4e$ ) step deprotonates the  $^*OOH$ , leading to state 5. Although PBE predicts a free energy barrier of  $\sim 0.4$  eV for state 5 to release  $O_2$  and recover the original state 1, B3PW91 finds

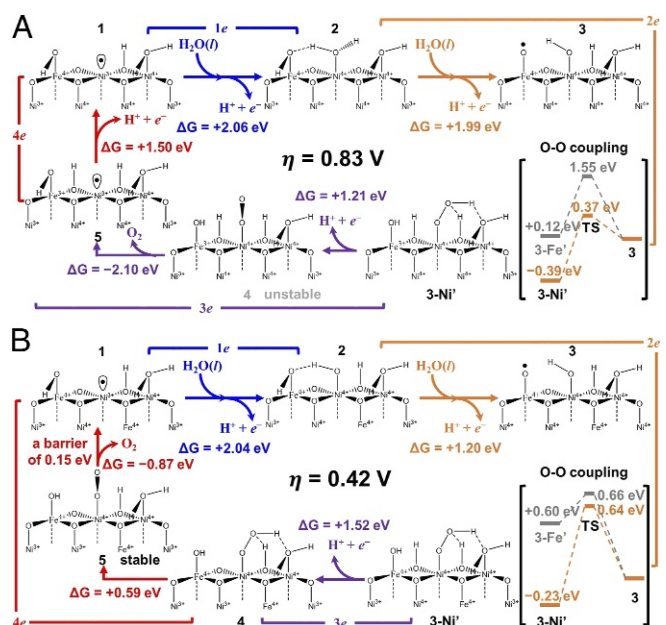
a barrierless downhill reaction, indicating an unstable state 5 and thus immediate  $O_2$  release after  $^*OOH$  deprotonation. This drives state 4' back to state 1. Thus, the  $4e$  step completes the OER cycle.

- Calculating the current density ( $j$ ) vs.  $U$

Having elucidated the four electron steps and the associated energetics (including barriers), we can now compute the current density ( $j$ ) at any  $U$ . We use a microkinetic model assuming the steady-state approximation (more details are in SI Appendix, SI Methods). Thus, we solve numerically for the onset potential  $U_{on}$  at which the  $U$ -dependent free energies and barriers deliver a  $j$  of 10 mA/cm<sup>2</sup>. To calculate the overpotential  $\eta = U_{on} - U_0$ , we use the equilibrium potential  $U_0$  of OER predicted by the corresponding level of theory (1.06 V by B3PW91, 1.13 V by PBE) for consistency. Both B3PW91 and PBE predict a much larger  $\eta$  of 1.22 and 0.97 V, respectively, than the experimental value of 0.7 V for OER on pure  $\gamma$ -NiOOH (10). This leads us to consider the role of Fe impurities. Indeed, some experimental studies have pointed out that it is very likely that preparation of NiOOH catalysts may easily have introduced Fe impurities accidentally (33).

**The Reaction Pathway of OER for 1/3 Fe on the Top Surface.** We first consider a single Fe dopant atom replacing one of the 12 Ni atoms among the four layers in our model. We find that this Fe very strongly prefers to substitute the surface  $Ni^{4+}$  site that has the  $^*OH$  (Fig. 3A, state 1). It is 0.4–0.5 eV higher in energy to move this Fe into subsurface Ni sites, making it likely that for small concentrations of Fe, all are at the surface. Indeed, even with trace amounts of Fe impurities, the surface Fe concentration may possibly reach as high as the fraction 1/3 considered here.

With the presence of 1/3 surface Fe doping in  $\gamma$ -NiOOH, we find a dramatic change in the OER mechanism (Fig. 3A). The key change is that the formation of active O radical is moved to the  $2e$  step, and the O radical is now bonded to the surface



**Fig. 3.** The mechanistic cycles for OER on  $\gamma$ -(Ni,Fe)OOH catalysts (based on B3PW91 predicted free energies and barriers). (A) One-third surface Fe doping case, leading to  $U = 1.89$  V ( $\eta = 0.83$  V) for  $j = 10$  mA/cm<sup>2</sup>. (B) One-third surface Fe doping plus 1/3 subsurface Fe doping, leading to  $U = 1.48$  V ( $\eta = 0.42$  V) for  $j = 10$  mA/cm<sup>2</sup>.



calculated Tafel slope of 61 mV/dec differs from the experiments (90–120 mV/dec), we conclude that the TLS for OER on the pure NiOOH is the chemical O–O coupling step after one reversible electron transfer step that generates the O radical right from the resting state. This scenario is similar to OER on the Co–Pi catalyst (28). The discrepancy between our prediction and experiments might arise from our assumption that the electron transfer is fast and not rate-limiting, while the experimental Tafel slopes of 90–120 mV/dec for OER on the pure NiOOH indicates an electron transfer step as the TLS.

Therefore, in addition to the synergy between Ni and Fe, the stabilization of O radical by Fe plays a second essential role in shifting the TLS to become the chemical O–O coupling preceded by two electron transfer steps, which leads to the much lower calculated Tafel slope of 23 mV/dec. This in turn allows only a small increase in applied potential to substantially increase the current.

**PBE vs. B3PW91 Flavor of DFT.** Our preceding discussion used the results from the B3PW91 flavor of DFT. This is because it is well known that such hybrid methods often lead to much more accurate results than PBE. We also carried out PBE calculations for all of the steps discussed for B3PW91. Table 1 summarizes the predicted  $\eta$ 's by B3PW91 and PBE for the pure and Fe-doped  $\gamma$ -NiOOH. Most interesting here is that PBE predicts that Fe doping leads to only slight improvement of OER performance, changing from  $\eta = 0.97$  to  $0.88$ – $0.79$  V for the cases in which 0, 1, and 2 of the 12 Ni are replaced with Fe. Thus, for OER, PBE leads to direct contradiction to experiment. In contrast, B3PW91 leads to  $\eta = 1.22$  to  $0.83$  to  $0.42$  V for the cases in which 0, 1, and 2 of the 12 Ni are replaced with Fe. Thus, B3PW91 is quite consistent with the dramatic improvement in performance observed experimentally. This highlights the necessity for using

hybrid functional methods to describe chemistry on catalysts with localized electrons/spins.

## Summary

Using advanced QM methods (hybrid DFT and GC-QM), we determined the atomistic mechanisms including the kinetics of O–O coupling for OER on pure and Fe-doped NiOOH catalysts. We discovered two key steps of OER:

- The formation of an active O radical species; and
- The subsequent O–O coupling.

Our results find that the pure NiOOH is a poor OER catalyst, because it is ineffective at producing the O radical. In contrast, the (Ni,Fe)OOH systems feature synergy between Fe and Ni: The high spin  $d^4$  Fe(IV) stabilizes the O radical by exchange interaction to generate it efficiently, while the closed shell  $d^6$  Ni(IV) catalyzes the O–O coupling. Thus, the two components assume the two different key functionalities needed for OER. It is this synergy that delivers the optimal performance of (Ni,Fe)OOH for driving the OER. Moreover, the stabilization of O radical by Fe plays an essential role in shifting the TLS to be the chemical O–O coupling preceded by two electron transfer steps, which leads to a much lower Tafel slope. This suggests that the (Ni,Fe)OOH-type OER catalysts might be improved further by separately screening for the best component for each key functionality.

**ACKNOWLEDGMENTS.** This work was supported by the Joint Center for Artificial Photosynthesis, a Department of Energy (DOE) Energy Innovation Hub, supported through the Office of Science of the DOE under Award DE-SC0004993. The calculations were carried out on the Extreme Science and Engineering Discovery Environment, which is supported by National Science Foundation Grant ACI-1053575, and the Zwicky Astrophysics Supercomputer at Caltech.

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